



The Role of Surface X-ray Scattering in Electrocatalysis

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Ecological and political realities have moved discussions of, and advances in, fuel cell technology into mainstream public awareness. Electrocatalysis, the science of modifying the overall rates of electrochemical reactions so that selectivity, yield and efficiency are maximized, is the work from which those advances spring. Studies in electrocatalysis have resulted in highly selective multicomponent gas mixture sensors, human blood component sensors, new electrocatalysts for oxidation/reduction of inorganic and organic pollutants in air and water, as well as better electrocatalysts for the fuel cell conversion of renewable and fossil fuels to electrical work. Studies of the mechanisms by which these catalysts operate have been advanced through development of *in-situ* surface x-ray scattering (SXS) techniques. SXS capabilities at SSRL were recently used to investigate the interface structure of an ultrathin CO_{ad} (adsorbed carbon monoxide) overlayer on platinum. This work has elevated the macroscopic description of the CO_{ad} state at the solid-liquid interface to a microscopic level and enabled the relation between the reactivity and the interfacial structure of CO_{ad}/Pt to be understood.

Of the various systems studied, CO oxidation [$CO_{ad} + OH_{ad} = CO_2 + H^+ + e^-$] on Pt and Pt-bimetallic surfaces occupies a special position in surface electrochemistry. Carbon monoxide is the simplest C₁ molecule that can be electrochemically oxidized in a low temperature fuel cell at a reasonable (although not necessarily practical) potential. It thus serves as an important model “fuel” for fundamental studies of C₁ electrocatalysis. For over a decade now the ability to characterize atomic/molecular spatial structures and to monitor changes in the local symmetry of surface atoms *in-situ* under reaction conditions has played an important part in our understanding of surface electrochemistry at metal-based interfaces [1,2]. This progress has been influenced greatly by the technique of *in-situ* surface x-ray scattering (SXS) which, in combination with infrared spectroscopy (IR) and electrochemical methods, has been used to find interrelationships between the microscopic surface structures of fcc metals (Pt, Ir, Pd, Au, Ag, Cu) and the macroscopic kinetic rates of the reactions. Two forms of CO_{ad} species can be distinguished thermochemically on Pt(111) in an electrochemical environment [3]: (i) CO_{ad} with a low heat of adsorption is characterized as the “weakly adsorbed” state, and (ii) CO_{ad} with a relatively high enthalpy of adsorption is characterized as the “strongly adsorbed” state. In our recent studies, the macroscopic description of the CO_{ad} state at the solid-liquid interface has been elevated to a microscopic level, which enables the relation between the reactivity and the interfacial structure of CO_{ad} on Pt(111) to be understood.

Representative SXS results along with CO oxidation current in 0.05 M H₂SO₄ are summarized in Figure 1. Direct information regarding the induced relaxation of surface Pt atoms by the adsorption of CO are obtained by analyzing and modeling crystal truncation rod (CTR) data [4] (not shown). The potential dependence of the Pt(111) surface relaxation, denoted as x-ray voltammetry (XRV) induced by CO_{ad} adsorption is represented by the results in Figure 1b. In CO_{ad}-free solution at 0.05 V the top Pt atomic layer expands

ca. $\approx 2\%$ (0.05 \AA) of the lattice spacing away from the second atomic layer when the adsorbed hydrogen (H_{ad}) reaches its maximum coverage (0.66 ML).

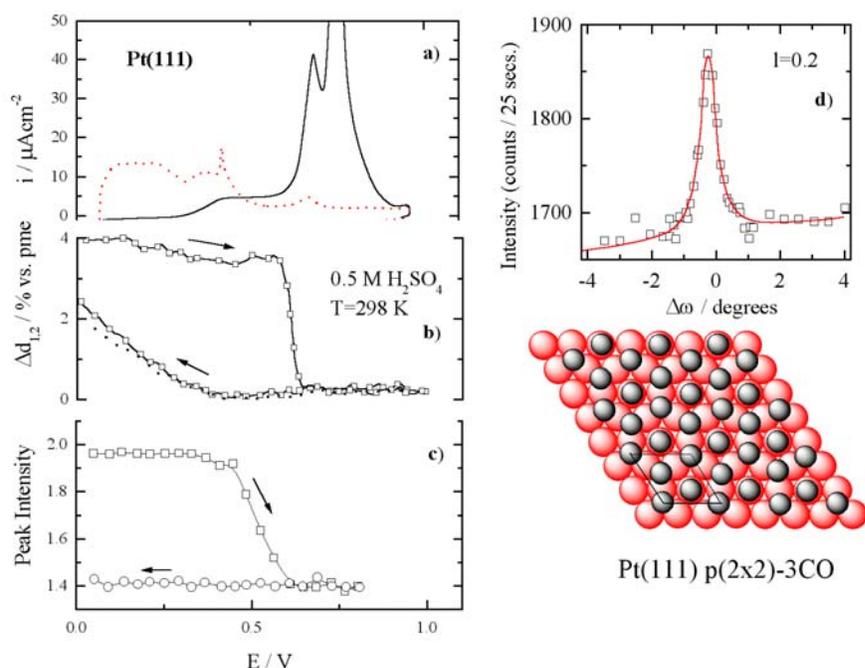


Figure 1. (a) CO stripping voltammetry on the Pt(111) surface in argon purged solution. (b) (—) Scattering intensity changes at $(1, 0, 3.6)$ for the surface covered by CO_{ad} ; (c) Measured x-ray intensity at $(1/2, 1/2, 0.2)$ as a function of electrode potential for the same conditions as in (b). (d) A rocking scan through the $(1/2, 1/2, 0.2)$ position and ideal model for the $p(2 \times 2)$ -3CO structure.

Following the adsorption of CO at 0.05 V, the Pt surface expansion is even larger ca. 4%. The difference in relaxation of the Pt(111) surface covered with H_{ad} and CO_{ad} probably arises from the difference in the adsorbate-metal bonding, the Pt(111)- CO_{ad} interaction being much stronger than the Pt(111)- H_{ad} interaction. At 0.05 V, no change in the relaxation of the Pt surface atoms was observed after replacement of CO from solution with nitrogen, indicating that CO_{ad} is indeed irreversibly adsorbed on the Pt(111) surface. Upon sweeping the potential positively from 0.05 V, the oxidation of CO_{ad} in the so-called pre-oxidation potential region ($0.3 < E < 0.6\text{ V}$) is mirrored with a small contraction of the Pt surface layer. Above ca. 0.6V, the top layer expansion is reduced significantly, contracting above 0.7 V to the unrelaxed state that the Pt(111) surface has in the absence of CO_{ad} .

Direct information regarding the CO_{ad} structure on Pt(111) was obtained by searching in the surface plane of reciprocal space for diffraction peaks characteristic of an ordered adlayer. At 0.05 V, a diffraction pattern consistent with a $p(2 \times 2)$ symmetry was observed between $0.05 < E < 0.6\text{ V}$. A rocking scan through the $(\frac{1}{2}, \frac{1}{2}, 0.2)$ position together with the derived structural model, which consists of three CO molecules per $p(2 \times 2)$ unit cell, is shown schematically in Figure 1d. From the width of this peak and from the result of similar fits to other $p(2 \times 2)$ reflections a coherent domain size in the range of 80-120 \AA for the CO adlayer was deduced. Upon the reversal of the electrode potential at ca. 0.6 V, the $p(2 \times 2)$ -3CO structure is not re-formed, Figure 1c, confirming that the structure is coverage-dependent and not just potential-dependent.

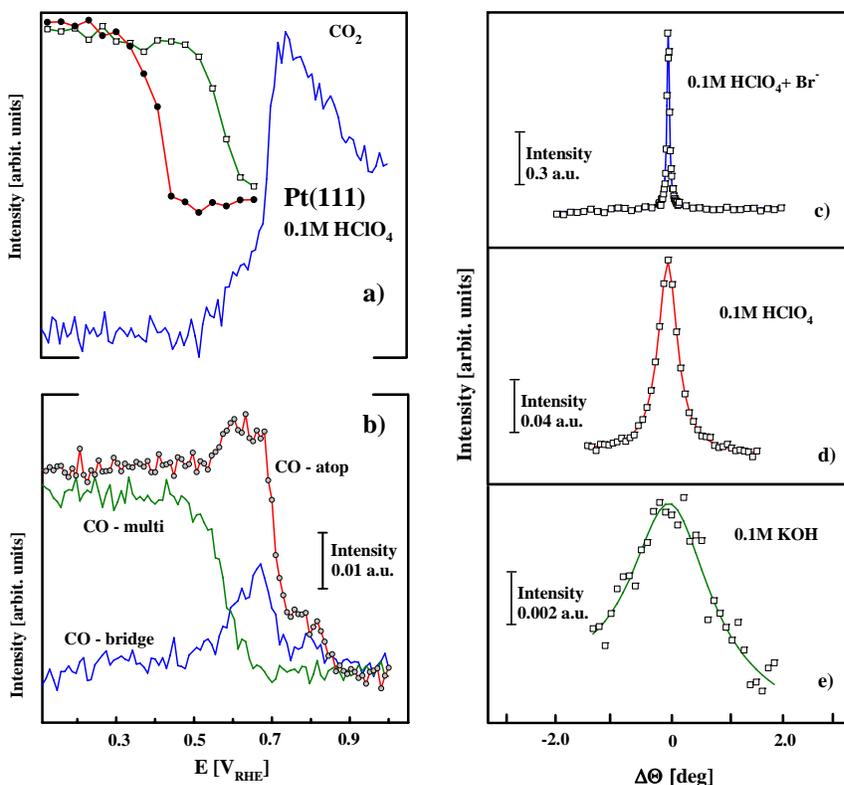


Figure 2. (a) Potential-dependent stability of the $p(2 \times 2)$ - 3CO_{ad} structure on Pt(111) in 0.1 M HClO_4 in CO-saturated solution, and CO_2 production as a function of electrode potential (data extracted from FTIR measurement) during the oxidation of CO_{ad} . (b) Integrated intensities for CO_{ad} -atop, CO_{ad} -multi and CO_{ad} -bridge on Pt(111) as a function of electrode potential in CO-saturated 0.1 M HClO_4 solution. A rocking scan through the $(1/2, 1/2, 0.2)$ position for the $p(2 \times 2)$ - 3CO_{ad} structure formed at 0.05 V Pt(111) in: (c) 0.1 M $\text{HClO}_4 + 10^{-3}$ M Br^- ; (d) 0.1 M HClO_4 ; and (e) 0.1 M KOH .

With a constant overpressure of CO in the x-ray cell, the SXS experiments revealed a *reversible* loss and re-formation of the $p(2 \times 2)$ - 3CO structure, with the $p(2 \times 2)$ - 3CO structure re-forming as the potential was slowly (1 mV/s) swept below 0.2 V, Figure 2a. IR has provided a valuable complement to the structural information obtained from SXS [3]. As shown in Figure 2b, the spectra for CO_{ad} on Pt(111) in CO-saturated 0.1 M HClO_4 have three characteristic Pt-CO stretching frequencies, the atop CO near 2070 cm^{-1} , the multi-coordinated CO_{ad} near 1780 cm^{-1} , and the bridge CO_{ad} near 1840 cm^{-1} . Figure 2 shows that oxidation of the CO_{ad} in the three-fold hollow sites and relaxation of the remaining CO_{ad} into bridge sites and atop sites is accompanied with both the decrease in the Bragg peak intensity for the $p(2 \times 2)$ - 3CO structure and CO_2 formation. Therefore, the macroscopic characterization of a weakly adsorbed state is linked microscopically to a saturated CO adlayer consisting of three CO_{ad} molecules per $p(2 \times 2)$ unit cell located in a-top and three-fold hollow sites of the Pt(111) surface. Figure 2 shows that oxidation of the CO_{ad} in the three-fold hollow sites and relaxation of the remaining CO_{ad} into bridge sites and atop sites disrupts the long-range ordering in the remaining adlayer, as the Bragg peak intensity for the $p(2 \times 2)$ - 3CO structure decreases rapidly in this potential region. The “relaxed” CO_{ad} adlayer (characterized microscopically) with substantial alternation in binding site geometry from predominantly three-fold hollow to bridge sites can be linked macroscopically to the strongly adsorbed state of CO_{ad} that is oxidized in the ignition potential region. It is also worth mentioning that besides tuning the fine balance between atop and multifold coordinated CO_{ad} , the domain size of the CO_{ad} structure is significantly affected by the nature of anions. For example, the domain size (Figures 2d-2f) and stability (not shown) of the $p(2 \times 2)$ - 3CO structure increases from KOH (ca. 30 \AA), to HClO_4 (ca. 140 \AA) to $\text{HClO}_4 + \text{Br}^-$ (ca. 350 \AA); *i.e.*, the less active the surface is towards CO_{ad} oxidation, the larger the ordered domains of the $p(2 \times 2)$ - 3CO structure. As discussed in reference [3], a self-consistent explanation for this result is that both the stability and domain size of the

ordered CO_{ad} adlayer are determined by the competition between OH_{ad} and spectator anions for the defect/step sites on the Pt(111) surface.

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