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## Advanced EXAFS analysis and considerations: What's under the hood

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### Topics

1. Overview
2. Theory
  - A. Simple "heuristic" derivation
  - B. polarization: oriented vs spherically averaged (won't cover)
  - C. Thermal effects
3. Experiment: Corrections and Other Problems
  - A. Sample issues (size effect, thickness effect, glitches) (won't cover)
  - B. Fluorescence mode: Dead time and Self-absorption (won't cover)
  - C. Energy resolution (won't cover)
4. Data Analysis
  - A. Correlations between parameters (e.g, mean-free path)
  - B. Fitting procedures (After this talk)
  - C. Fourier concepts
  - D. Systematic errors (won't cover)
  - E. "Random" errors (won't cover)
  - F. F-tests

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### X-ray absorption spectroscopy (XAS) experimental setup

- sample absorption is given by  $\mu t = \log(I_0/I_f)$
- reference absorption is  $\mu_{\text{REF}} t = \log(I_0/I_1)$
- sample absorption in fluorescence  $\mu \propto I_F/I_0$
- NOTE: because we are always taking relative-change ratios, detector gains don't matter!

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### X-ray absorption spectroscopy

- Main features are single-electron excitations.
- Away from edges, energy dependence fits a power law:  $\mu \propto E^{-3} + BE^{-4}$  (Victoreen).
- Threshold energies  $E_0 \sim Z^2$ , absorption coefficient  $\mu \sim Z^4$ .

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### X-ray absorption fine-structure (XAFS) spectroscopy

"edge region": x-ray absorption near-edge structure (XANES)  
near-edge x-ray absorption fine-structure (NEXAFS)

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### Data reduction: $\mu(E) \rightarrow \chi(k, r)$

$\chi(k) = \frac{\mu - \mu_0}{\mu_0}$

$\hbar^2 k^2 = E - E_0$

$2m_e$

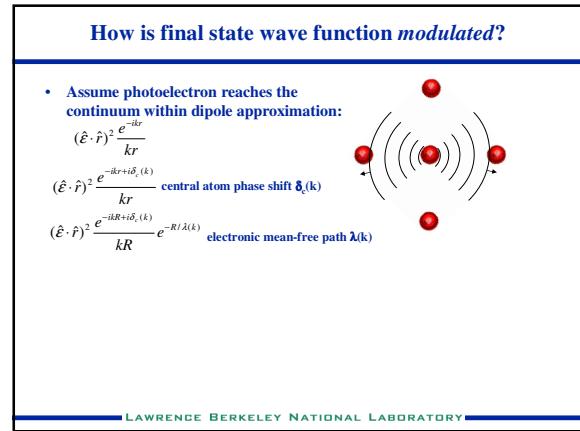
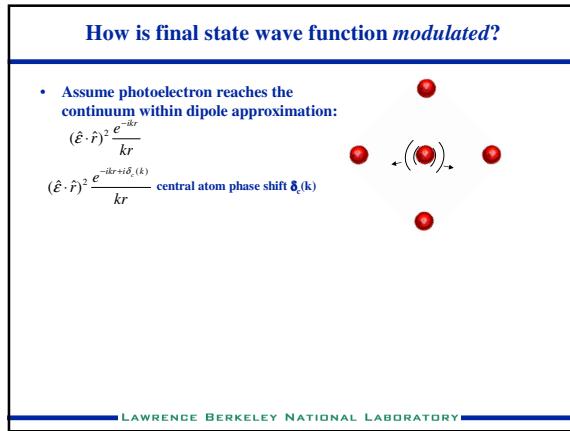
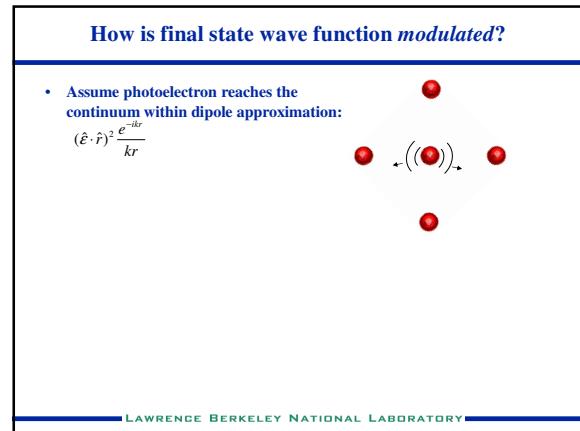
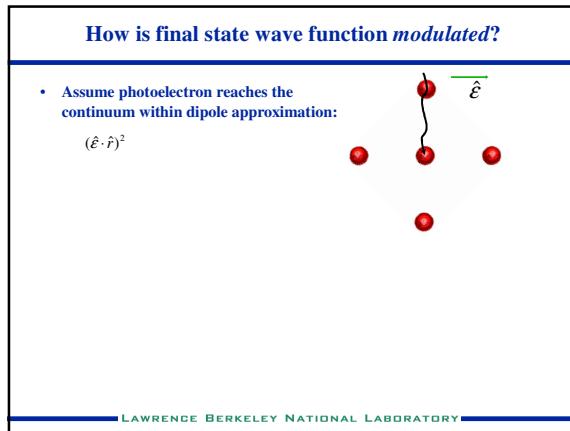
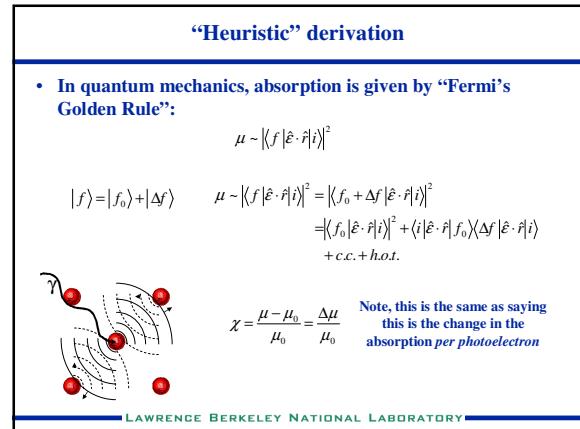
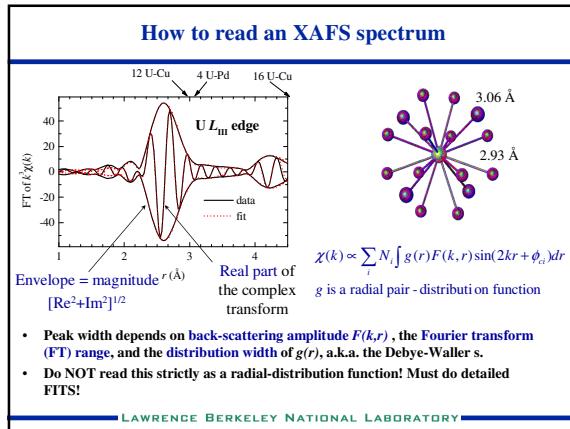
$k = 0.512\sqrt{E - E_0}$

$\chi''(E)$

$k(\text{\AA}^{-1})$

U  $L_{3,2}$  edge

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### How is final state wave function modulated?

- Assume photoelectron reaches the continuum within dipole approximation:

$$(\hat{E} \cdot \hat{r})^2 \frac{e^{-ikr}}{kr}$$

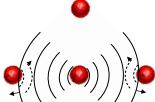
central atom phase shift  $\delta_c(k)$

$$(\hat{E} \cdot \hat{r})^2 \frac{e^{-ikr+i\delta_c(k)}}{kr} e^{-R/\lambda(k)}$$

electronic mean-free path  $\lambda(k)$

$$(\hat{E} \cdot \hat{r})^2 \frac{e^{-ikR+i\delta_c(k)}}{kR} kf(\pi, k) e^{-R/\lambda(k)}$$

complex backscattering probability  $f(g, k)$



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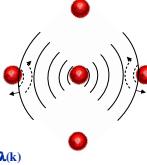
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complex backscattering probability  $f(g, k)$

$$(\hat{E} \cdot \hat{r})^2 \frac{e^{-ikR+i\delta_c(k)}}{kR} k|f(\pi, k)| e^{-R/\lambda(k)} \frac{e^{-ik(R-r)+i\delta_c(k)+i\delta_a(k)}}{kr}$$

complex=magnitude and phase:  
backscattering atom phase shift  $\delta_a(k)$



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### How is final state wave function modulated?

- Assume photoelectron reaches the continuum within dipole approximation:

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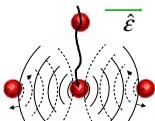
complex backscattering probability  $kf(g, k)$

$$(\hat{E} \cdot \hat{r})^2 \frac{e^{-ikR+i\delta_c(k)}}{kR} k|f(\pi, k)| e^{-R/\lambda(k)} \frac{e^{-ik(R-r)+i\delta_c(k)+i\delta_a(k)}}{kr}$$

complex=magnitude and phase:  
backscattering atom phase shift  $\delta_a(k)$

$$\text{Im}(\hat{E} \cdot \hat{r})^2 \frac{e^{-ikR+i\delta_c(k)+i\delta_a(k)}}{kr^2} |f(\pi, k)| e^{2R/\lambda(k)}$$

final interference modulation per point atom!



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### Other factors

- Allow for multiple atoms  $N_i$  in a shell  $i$  and a distribution function of bondlengths within the shell  $g(r)$

$$\chi(k) = S_0^2 \sum_i N_i \int (\hat{E} \cdot \hat{r})^2 |f(\pi, k)| e^{-2r/\lambda(k)} g(r) \frac{\sin 2kr + 2\delta_c(k) + \delta_a(k)}{kr^2} dr$$

where  $g(r) = \frac{1}{\sqrt{2\pi}\sigma} e^{-\frac{(r-R_i)^2}{2\sigma^2}}$  and  $S_0^2$  is an inelastic loss factor

$$\chi(k) = S_0^2 \sum_i N_i (\hat{E} \cdot \hat{r})^2 |f(\pi, k)| e^{-2r/\lambda(k)} \frac{e^{-2k^2\sigma^2} \sin 2kr + 2\delta_c(k) + \delta_a(k)}{kr^2}$$

Requires curved wave scattering, has  $r$ -dependence, use full curved wave theory:  
FEFF

Assumed both harmonic potential AND  
 $k\sigma \ll 1$ : problem at high  $k$  and/or  $\sigma$  (good to  
 $k\sigma$  of about 1)

### Some words about Debye-Waller factors

$$\chi(k) = S_0^2 \sum_i N_i \int (\hat{E} \cdot \hat{r})^2 |f(\pi, k)| e^{-2r/\lambda(k)} g_i(r) \frac{\sin 2kr + 2\delta_c(k) + \delta_a(k)}{kr^2} dr$$

- Harmonic approximation: Gaussian

$$g_i(r) = \frac{1}{\sqrt{2\pi}\sigma} e^{-\frac{(r-r_i)^2}{2\sigma^2}}$$

(non-Gaussian is advanced topic:  
“cumulant expansion”)

$$\sigma_{ij}^2 = \langle (\Delta P_i - \Delta P_j)^2 \rangle = \langle \Delta P_i^2 + \Delta P_j^2 - 2\Delta P_i \Delta P_j \rangle$$

$$= \langle \Delta P_i^2 \rangle + \langle \Delta P_j^2 \rangle - 2\langle \Delta P_i \Delta P_j \rangle$$

$$\sigma_{ij}^2 = U_i^2 + U_j^2 - 2\phi U_i U_j$$

- $U_i^2$  are the position mean-squared displacements (MSDs) from diffraction

- Important: EXAFS measures MSD differences in position  
(in contrast to diffraction!)

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### Lattice vibrations and Debye-Waller factors

- isolated atom pair, spring constant  $\kappa$

$$m_r = (1/m_i + 1/m_j)^{-1}$$

Classically...

$$E = \frac{1}{2} m_r v^2 + \frac{1}{2} \kappa x^2$$

$$< E > = < n + \frac{1}{2} > \hbar \omega$$

$$< E > = < \frac{1}{2} m_r v^2 + \frac{1}{2} \kappa x^2 >$$

$$= < \kappa x^2 > = < x^2 > = \kappa \sigma^2$$

$$\text{with } \kappa = m_r \omega^2$$

Quantumly...

$$< E > = < n + \frac{1}{2} > \hbar \omega$$

$$< E > = k_B T \text{ at high } T$$

$$\sigma^2 = \frac{< E >}{\kappa} \approx \frac{k_B T}{\kappa}$$

$$\sigma^2 = \frac{\hbar \omega}{\kappa} \text{ at low } T$$

PHI TERM ABOVE is +1 IF ATOM MOTIONS ARE PERFECTLY CORRELATED,  
0 IF THEY ARE UNCORRELATED AND -1 IF ANTICORRELATED

## Some words about Debye-Waller factors

- The general formula for the variance of a lattice vibration is:

$$\sigma_{\text{v}}^2 = \frac{\hbar}{2m} \int \frac{d\omega}{\omega} \rho_p(\omega) \coth\left(\frac{\hbar\omega}{2k_B T}\right)$$

where  $\rho_p(\omega)$  is the projected density of modes with vibrational frequency  $\omega$

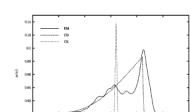
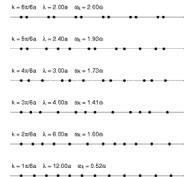


FIG. 1. Projected DOS  $\rho_p(\omega)$  for the first shell of Cu sites, band with the EM method with  $N=459$  and  $k=27.9$  Nm/Å.

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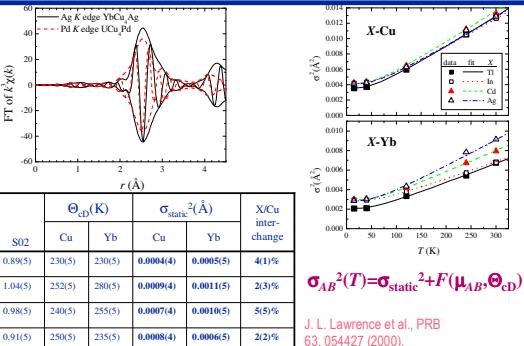
- Einstein model: single frequency
- correlated-Debye model: quadratic and linear dispersion

$\Theta_{\text{CD}} = c k_{\text{CD}}$

Polarkova and Rehr, Phys. Rev. B 59, 948 (1999).

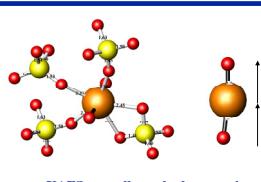
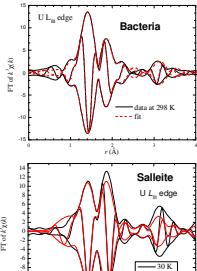
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## A “zero-disorder” example: YbCu<sub>4</sub>X



J. L. Lawrence et al., PRB 63, 054427 (2000).

## XAFS of uranium-bacterial samples



- XAFS can tell us whether uranium and phosphate form a complex
- U-O<sub>ax</sub> are very stiff  $\Theta_{\text{CD}} \sim 1000$  K
- U-O<sub>ax</sub>-O<sub>ax</sub> also stiff! Won't change with temperature
- U-P much looser  $\Theta_{\text{CD}} \sim 300$  K: Enhance at low T w.r.t. U-O<sub>ax</sub>-O<sub>ax</sub>!

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## Fitting the data to extract structural information

- Fit is to the standard EXAFS equation using either a theoretical calculation or an experimental measurement of  $F_{\text{eff}}$

$$F(k) = S_0^2 \sum_i N_i \int (\hat{e} \cdot \hat{r}_i)^2 |f_i(\pi, k)| e^{-r_i/\lambda_i} g_i(r) \frac{\sin 2kr + 2\delta_c(k) + \delta_s(k)}{kr^2} dr$$

- Typically, polarization is spherically averaged, doesn't have to be
- Typical fit parameters include:  $R_i$ ,  $N_i$ ,  $\sigma_i$ ,  $\Delta E_i$
- Many codes are available for performing these fits:
  - EXAFSPAK
  - IFEFFIT
    - SIXPACK
    - ATHENA
  - GNXAS
  - RSXAP

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## FEFF: a curved-wave, multiple scattering EXAFS and XANES calculator

- The FEFF Project is lead by John Rehr and is very widely used and trusted
- Calculates the complex scattering function  $F_{\text{eff}}(k)$  and the mean-free path  $\lambda$

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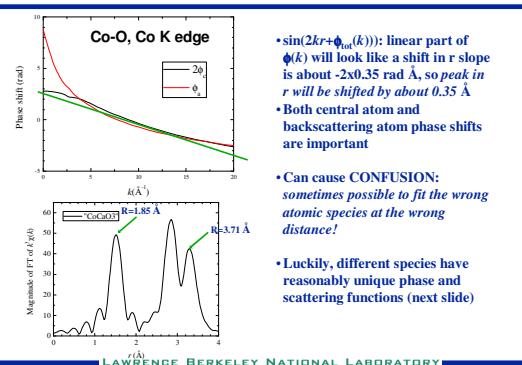
TITLE CaMnO3 from Poepelmeier 1982
HOLE 1 1.0 Mn K edge ( 6.540 keV), s0^2=1.0
POTENTIALS
* ipot z label
  0 22 Mn
  1 0
  2 20 Ca
  3 22 Mn

ATOMS
  0.00000 0.00000 0.00000 0 Mn 0.00000
  0.00000 -1.85615 0.00000 1 O(1) 1.85615
  0.00000 1.85615 0.00000 1 O(1) 1.85615
-1.31250 0.00000 1.31250 1 O(2) 1.85616
  1.31250 0.00000 -1.31250 1 O(2) 1.85616
-1.31250 0.00000 -1.31250 1 O(2) 1.85616
  1.31250 0.00000 1.31250 1 O(2) 1.85616
  0.00000 1.85615 -2.62500 2 Ca 3.21495
-2.62500 1.85615 0.00000 2 Ca 3.21495
-2.62500 -1.85615 0.00000 2 Ca 3.21495
  0.00000 1.85615 2.62500 2 Cs 3.21495

```

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## Phase shifts: functions of $k$

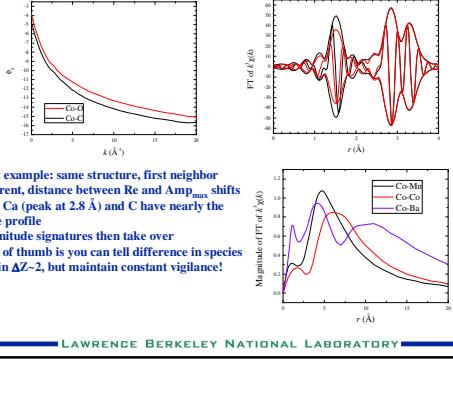


- $\sin(2kr + \phi_{\text{tot}}(k))$ : linear part of  $\phi(k)$  will look like a shift in  $r$ . slope is about  $-2\pi/0.35$  rad/Å, so peak in  $r$  will be shifted by about 0.35 Å
- Both central atom and backscattering atom phase shifts are important

- Can cause CONFUSION: sometimes possible to fit the wrong atomic species at the wrong distance!

- Luckily, different species have reasonably unique phase and scattering functions (next slide)

## Species identification: phase and magnitude signatures



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## More phase stuff: r and E<sub>0</sub> are correlated

- When fitting, ΔE<sub>0</sub> generally is allowed to float (vary)
  - In theory, a single ΔE<sub>0</sub> is needed for a monovalent absorbing species
  - Errors in ΔE<sub>0</sub> act like a **phase shift** and correlate to errors in R!
- consider error ε in E<sub>0</sub>:  $k_{\text{true}} = 0.512[E - (E_0 + \epsilon)]^{1/2}$   
 for small ε,  $k = k_0 \cdot [(0.512)^2 / (2k_0)] \epsilon$   
 e.g. at  $k = 10 \text{ Å}^{-1}$  and  $\epsilon = 1 \text{ eV}$ ,  $\Delta r \sim 0.013 \text{ Å}$
- This correlation is not a problem if k<sub>max</sub> is reasonably large
  - Correlation between N, S<sub>0</sub><sup>2</sup> and σ is a much bigger problem!*

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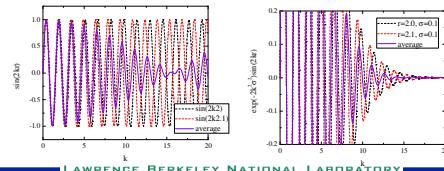
## Information content in EXAFS

- k-space vs. r-space fitting are *equivalent* if done correctly!
- 
- r-range in k-space fits is determined by scattering shell with highest R
  - k-space direct comparisons with raw data (i.e. residual calculations) are typically incorrect: must Fourier filter data over r-range
  - All knowledge from spectral theory applies! Especially, discrete sampling Fourier theory...

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## Fourier concepts

- highest “frequency”  $r_{\text{max}} = \pi/(2\delta_k)$  (Nyquist frequency)  
 e.g. for sampling interval  $\Delta k = 0.05 \text{ Å}^{-1}$ ,  $r_{\text{max}} = 31 \text{ Å}$
- for  $N_{\text{data}}$ , discrete Fourier transform has  $N_{\text{data}}$ , too! Therefore...  
 FT resolution is  $\delta_r = r_{\text{max}}/N_{\text{data}} = \pi/(2k_{\text{max}})$ , e.g.  $k_{\text{max}} = 15 \text{ Å}^{-1}$ ,  $\delta_r = 0.1 \text{ Å}$
- This is the ultimate limit, corresponds to when a beat is observed in two sine wave δ<sub>k</sub> apart. IF YOU DON’T SEE A BEAT, DON’T RELY ON THIS EQUATION!!



## More Fourier concepts: Independent data points

- Spectral theory indicates that each point in k-space affects every point in r-space. Therefore, assuming a fit range over k (and r):
- $$\Delta_r = r_2 - r_1 \quad \Delta_k = k_2 - k_1$$
- $$r_{\text{max}} = \frac{\pi}{2\delta_k} \quad N_k = \frac{k_{\text{max}}}{\delta_k} + 1 = 2 \left( \frac{r_{\text{max}}}{\delta_r} + 1 \right)$$
- $$N_{\text{ind}} = \frac{2}{\pi} \Delta_r \Delta_k + 2 + \text{h.o.t.} \text{ "Stern's rule" EXAFS result}$$
- Fit degrees of freedom  $v = N_{\text{ind}} - N_{\text{fit}}$
  - Generally should never have  $N_{\text{fit}} > N_{\text{ind}}$  ( $v < 1$ )  
 But what does this mean? It means that:  
 For  $N_{\text{fit}}$  exceeding  $N_{\text{ind}}$  there are other linear combinations of  $N_{\text{fit}}$  that produce EXACTLY the same fit function

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## not so Advanced Topic: F-test

- 
- F-test, commonly used in crystallography to test one fitting model versus another
- $$F = (\chi^2_{\text{true}}/\nu_1) / (\chi^2_{\text{obs}}/\nu_0) = \nu_0/\nu_1 \times R_0^2/R_{\text{obs}}^2$$
- (if errors approximately cancel)  
 alternatively:  $F = ((R_0^2 - R_{\text{obs}}^2)/(\nu_1 \nu_0)) / ((R_{\text{obs}}^2)/\nu_0)$
- Like χ<sup>2</sup>, F-function is tabulated, is given by incomplete beta function
  - Advantages over a χ<sup>2</sup>-type test:  
 —don’t need to know the errors!

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## Statistical tests between models: the F-test

$$F = \frac{\chi_1^2 / v_1}{\chi_0^2 / v_0} \quad R \propto \sqrt{\sum_i (y_i - y_i^f)^2}$$

$$F = \frac{(\chi_1^2 - \chi_0^2) / (v_1 - v_0)}{\chi_0^2 / v_0}$$

$$F = \left[ \left( \frac{R_1}{R_0} \right)^2 - 1 \right] \frac{(n-m)}{b}$$

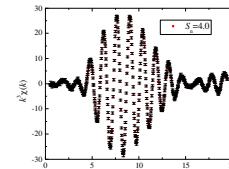
$$\alpha = P(F > F_{b,n-m,\alpha}) = 1 - I_{\mathcal{R}^{-1}} \left[ \frac{n-m}{2}, \frac{b}{2} \right]$$

where  $\alpha$  is the confidence level and  $I_x[y,z]$  is an incomplete beta function

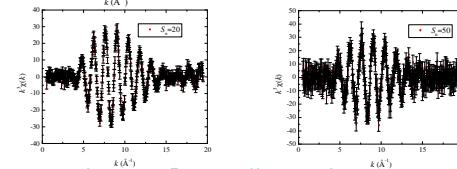
- F-tests have big advantages for data sets with poorly defined noise levels
  - with systematic error,  $R_1$  and  $R_0$  increase by ~constant, reducing the ratio (right direction!)

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## F-test example testing resolution in EXAFS: knowledge of line shape huge advantage

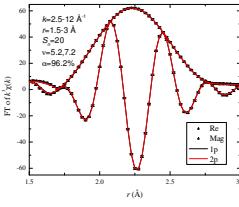


- 6 Cu-Cu at 2.55 Å, 6 at 2.65 Å
- $k_{\max} = \pi/(2\delta_R) = 15.7 \text{ \AA}^{-1}$
- 1 peak (+cumulants) vs 2 peak fits
- Fits to 18 Å⁻¹ all pass the F-test
- $S_n=50$  does not pass at 15 Å⁻¹
- $S_n=20$  does not pass at 13 Å⁻¹
- $S_n=4$  does not pass at ~11 Å⁻¹



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## Unfortunately, systematic errors dominate



Example without systematic error:  
 $k_{\max}=13 \text{ \AA}^{-1}, v_0=6.19, v_1=7.19$   
 $R_0=0.31$   
 $R_1=0.79$   
 $\alpha=0.999 \rightarrow \text{passes F-test}$

Example with systematic error at  $R=2\%$ :  
 $k_{\max}=13 \text{ \AA}^{-1}, v_0=6.19, v_1=7.19$   
 $R_0=0.31+2.0=2.31$   
 $R_1=0.79+2.0=2.79$   
 $\alpha=0.86 \rightarrow \text{does not pass}$

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## Finishing up

- Never report two bond lengths that break the resolution rule
- Break Stern's rule only with extreme caution
- Pay attention to the statistics

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## Further reading

- Overviews:
  - B. K. Teo, "EXAFS: Basic Principles and Data Analysis" (Springer, New York, 1986).
  - Hayes and Boyce, *Solid State Physics* 37, 173 (1982).
  - "X-Ray Absorption: Principles, Applications, Techniques of EXAFS, SEXAFS and XANES", ed. by Koningsberger and Prins (Wiley, New York, 1988).
- Historically important:
  - Sayers, Stern, Lytle, *Phys. Rev. Lett.* 71, 1204 (1971).
- History:
  - Lytle, *J. Synch. Rad.* 6, 123 (1999). (<http://www.exafsc.com/techpapers/index.html>)
  - Stumm von Bordwehr, *Ann. Phys. Fr.* 14, 377 (1989).
- Theory papers of note:
  - Lee, *Phys. Rev. B* 13, 5261 (1976).
  - Rehr and Albers, *Rev. Mod. Phys.* 72, 621 (2000).
- Useful links
  - xafs.org (especially see Tutorials section)
  - <http://www.i-x-s.org/> (International XAS society)
  - <http://www.csri.llnl.gov/periodic-table.html> (absorption calculator)

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## Further reading

- Thickness effect: Stern and Kim, *Phys. Rev. B* 23, 3781 (1981).
- Particle size effect: Lu and Stern, *Nucl. Inst. Meth.* 212, 475 (1983).
- Glitches:
  - Bridges, Wang, Boyce, *Nucl. Instr. Meth. A* 307, 316 (1991); Bridges, Li, Wang, *Nucl. Instr. Meth. A* 320, 548 (1992); Li, Bridges, Wang, *Nucl. Instr. Meth. A* 340, 420 (1994).
- Number of independent data points: Stern, *Phys. Rev. B* 48, 9825 (1993).
- Theory vs. experiment:
  - Li, Bridges and Booth, *Phys. Rev. B* 52, 6332 (1995).
  - Kvitky, Bridges, van Dorssen, *Phys. Rev. B* 64, 214108 (2001).
- Polarized EXAFS:
  - Heald and Stern, *Phys. Rev. B* 16, 5549 (1977).
  - Booth and Bridges, *Physica Scripta* T115, 202 (2005). (Self-absorption)
- Hamilton (F-test):
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### Further reading

- Correlated-Debye model:
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