

# **Thin Film Scattering: Epitaxial Layers**

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**First Annual SSRL Workshop on Synchrotron X-ray Scattering  
Techniques in Materials and Environmental Sciences: Theory and  
Application**

**Tuesday, May 16 & Wednesday, May 17, 2006**

- Thin films. Epitaxial thin films.
- What basic information we can obtain from x-ray diffraction
- Reciprocal space and epitaxial thin films
- Scan directions – reciprocal vs. real space scenarios
- Mismatch, strain, mosaicity, thickness
- How to choose right scans for your measurements
- Mosaicity vs. lateral correlation length
- SiGe(001) layers on Si(001) example
- Why sometimes we need channel analyzer
- What can we learn from reciprocal space maps
- SrRuO<sub>3</sub>(110) on SrTiO<sub>3</sub>(001) example
- Summary

## **What is thin film/layer?**

**Material so thin that its characteristics are dominated primarily by two dimensional effects and are mostly different than its bulk properties**

*Source: [semiconductorglossary.com](http://semiconductorglossary.com)*

**Material which dimension in the out-of-plane direction is much smaller than in the in-plane direction.**

**A thin layer of something on a surface**

*Source: [encarta.msn.com](http://encarta.msn.com)*

# Epitaxial Layer

**A single crystal layer that has been deposited or grown on a crystalline substrate having the same structural arrangement.**

*Source: photonics.com*

**A crystalline layer of a particular orientation on top of another crystal, where the orientation is determined by the underlying crystal.**

## Homoepitaxial layer

the layer and substrate are the same material and possess the same lattice parameters.

## Heteroepitaxial layer

the layer material is different than the substrate and usually has different lattice parameters.

## Thin films structural types

<b>Structure Type</b>	<b>Definition</b>
Perfect epitaxial	Single crystal in perfect registry with the substrate that is also perfect.
Nearly perfect epitaxial	Single crystal in nearly perfect registry with the substrate that is also nearly perfect.
Textured epitaxial	Layer orientation is close to registry with the substrate in both in-plane and out-of-plane directions. Layer consists of mosaic blocks.
Textured polycrystalline	Crystalline grains are preferentially oriented out-of-plane but random in-plane. Grain size distribution.
Perfect polycrystalline	Randomly oriented crystallites similar in size and shape.
Amorphous	Strong interatomic bonds but no long range order.

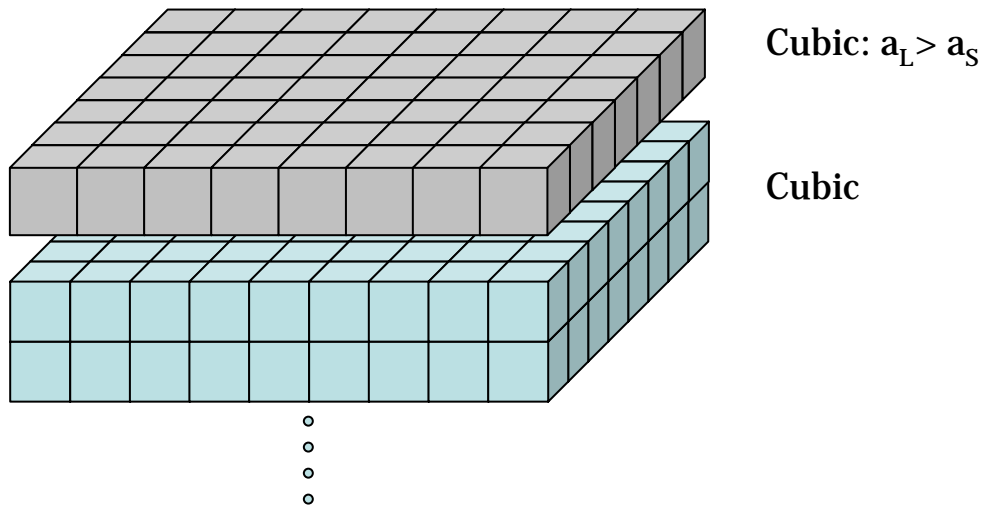
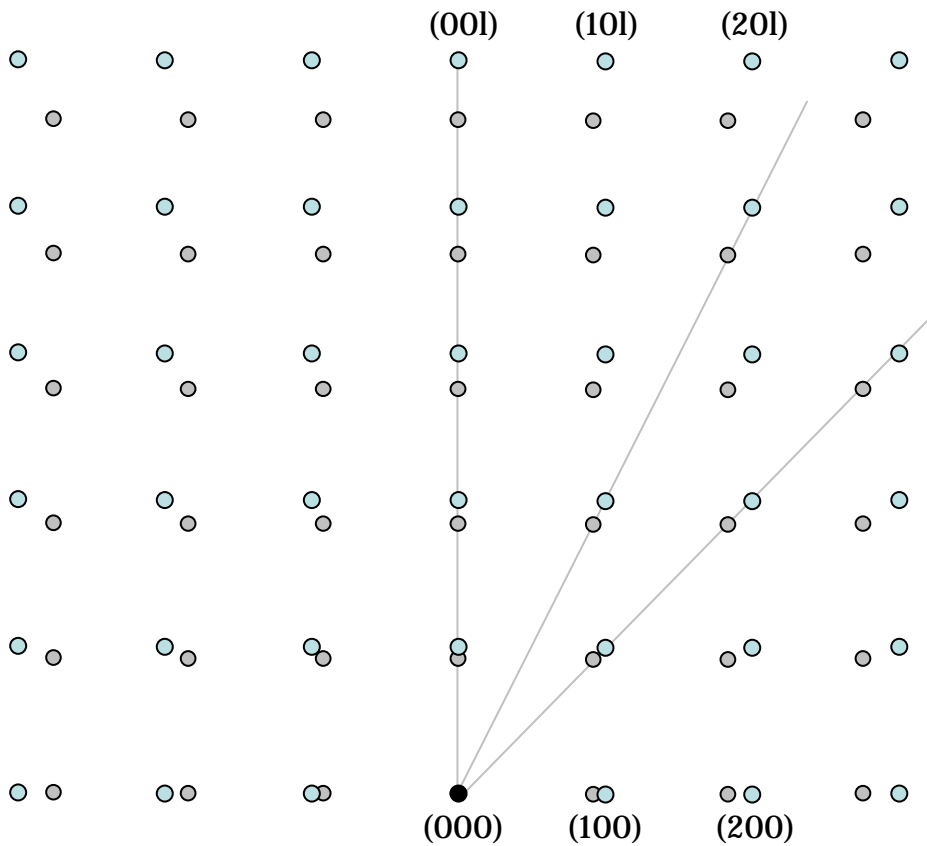
## **What we want to know about thin films?**

- **Crystalline state of the layers:**
  - Epitaxial (coherent with the substrate, relaxed)
  - Polycrystalline (random orientation, preferred orientation)
  - Amorphous
- **Crystalline quality**
- **Strain state (fully or partially strained, fully relaxed)**
- **Defect structure**
- **Chemical composition**
- **Thickness**
- **Surface and/or interface roughness**

# Overview of structural parameters that characterize various thin films

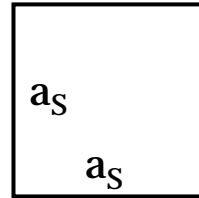
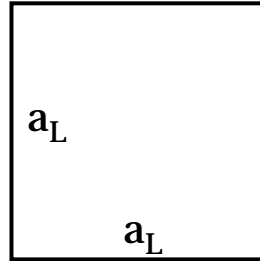
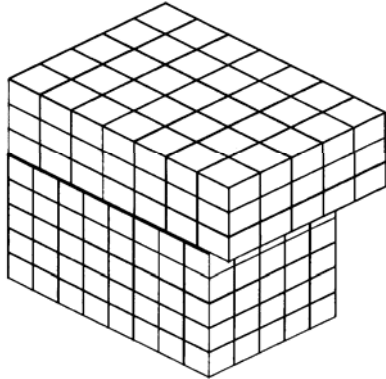
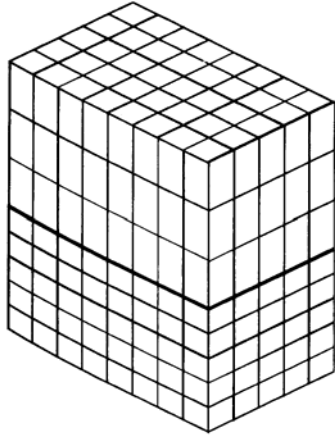
	Thickness	Composition	Relaxation	Distortion	Crystalline size	Orientation	Defects
Perfect epitaxy	×	×				×	
Nearly perfect epitaxy	×	×	?	?	?	×	×
Textured epitaxy	×	×	×	×	×	×	×
Textured polycrystalline	×	×	?	×	×	×	?
Perfect polycrystalline	×	×		×	×		?
Amorphous	×	×					

# Relaxed Layer

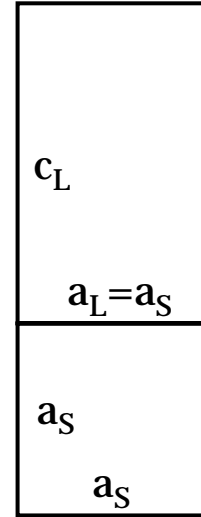




# Tetragonal Distortion



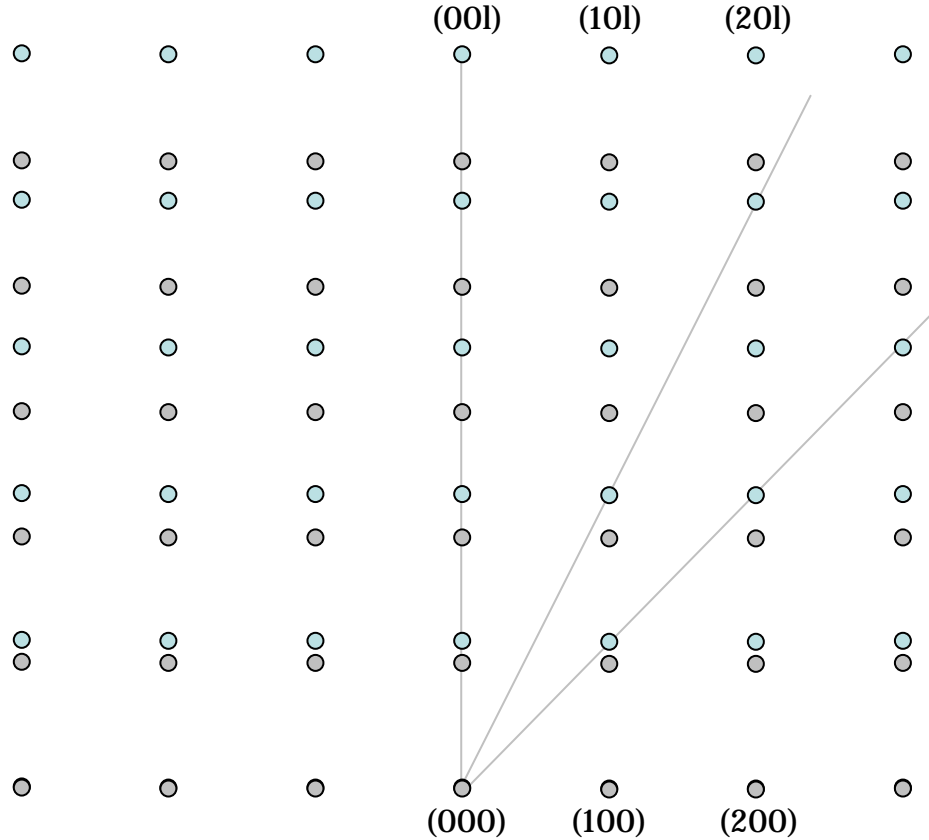
Before  
deposition



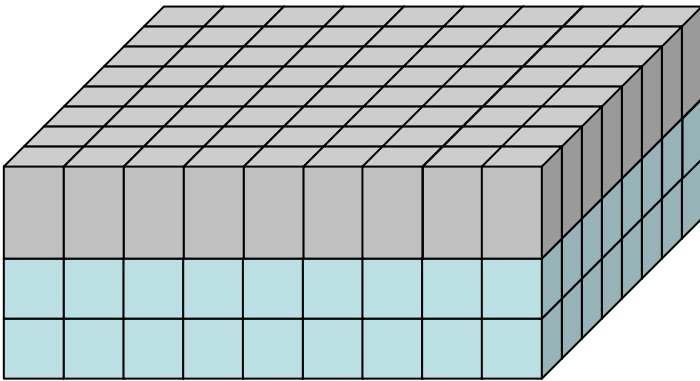
After  
deposition

$$\varepsilon_{\perp} = \varepsilon_{zz} = \frac{d_z^L - d_z^{L_0}}{d_z^{L_0}}$$

# Strained Layer



Tetragonal distortion →



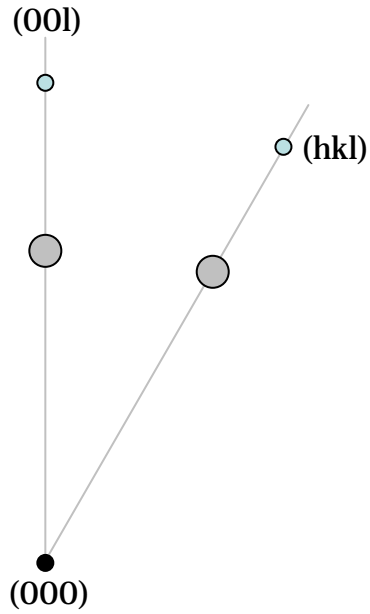
Tetragonal:  $a_{\parallel}^{\text{II}} = a_{\text{S}}, a_{\perp}^{\text{L}} > a_{\text{S}}$

Cubic

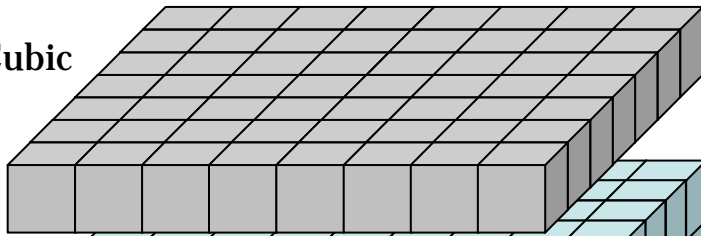
⋮

# Perfect Layers: Relaxed and Strained

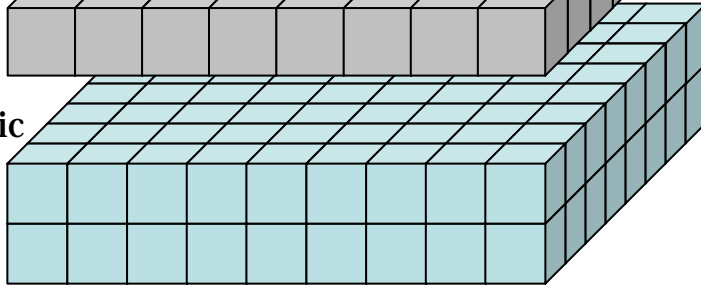
Reciprocal Space



Cubic

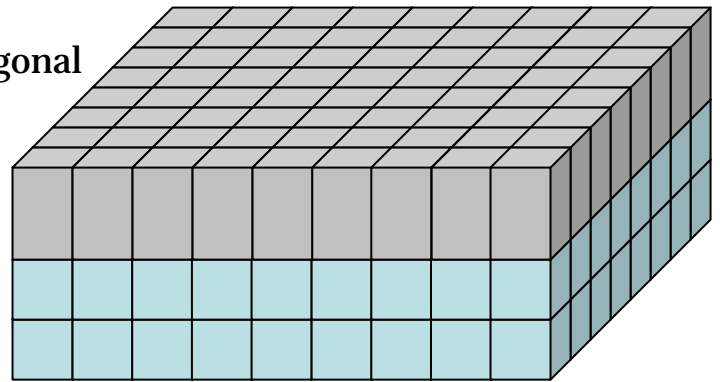


Cubic



$a_L > a_S$

Tetragonal

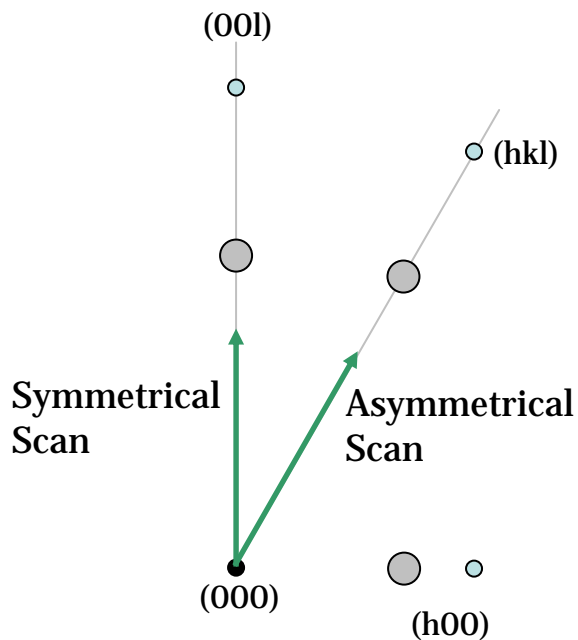
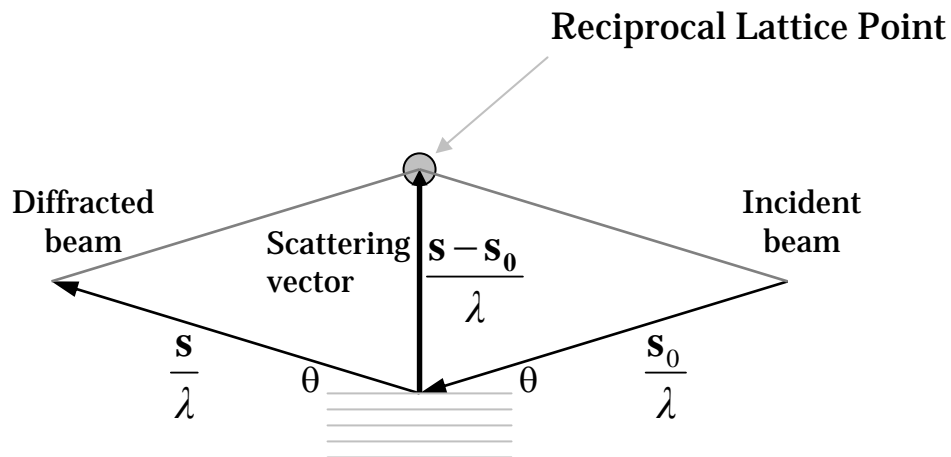


Cubic

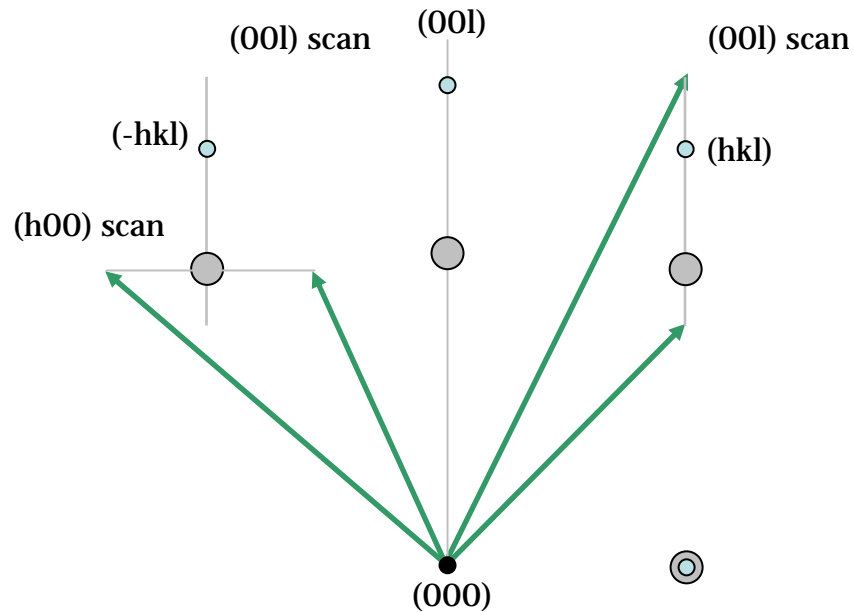
# Scan Directions

$$\left| \frac{\mathbf{s} - \mathbf{s}_0}{\lambda} \right| = \frac{2 \sin \theta}{\lambda} = \left| \mathbf{d}_{hkl}^* \right| = \frac{1}{d_{hkl}}$$

$$\lambda = 2d_{hkl} \sin \theta$$

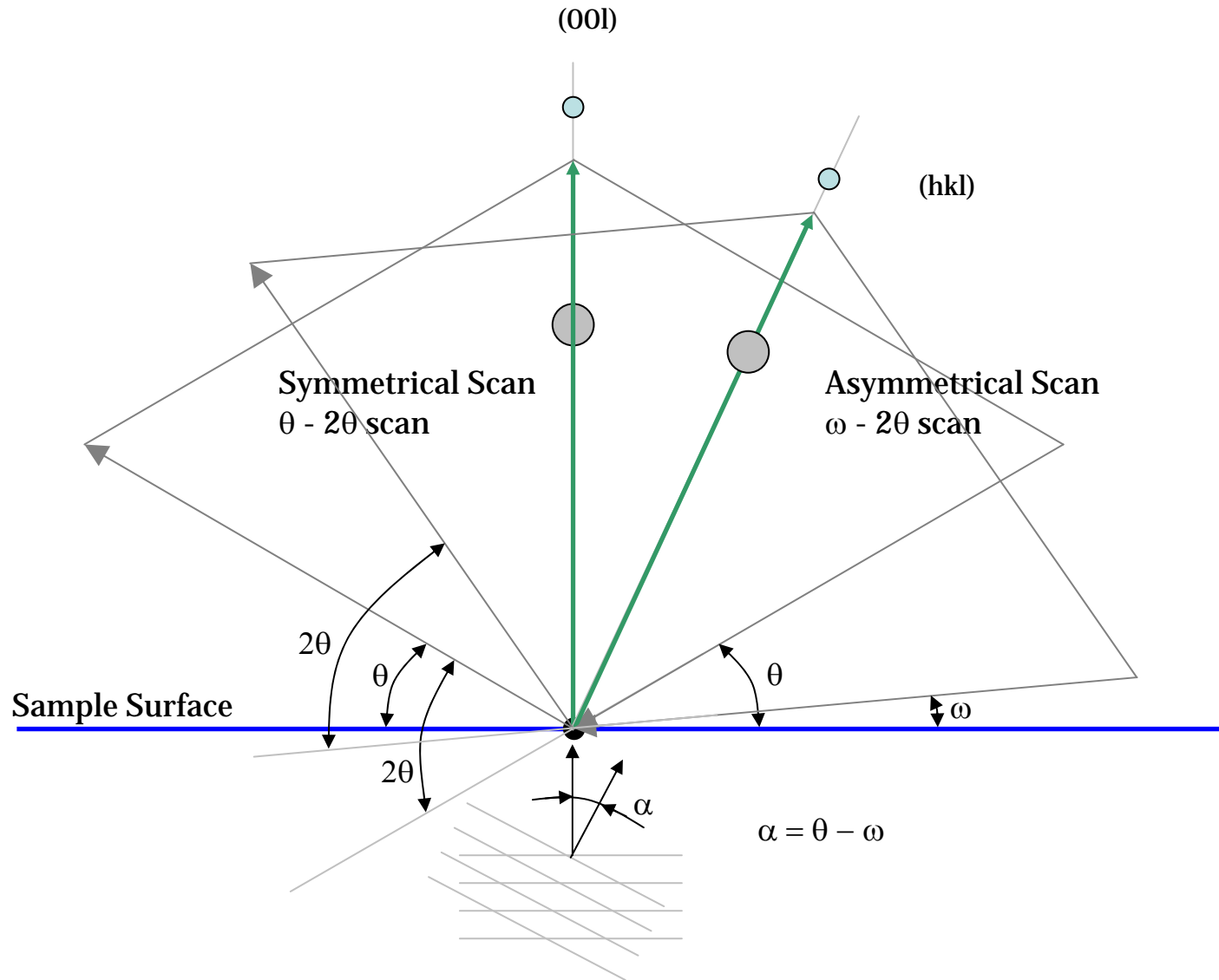


Relaxed Layer

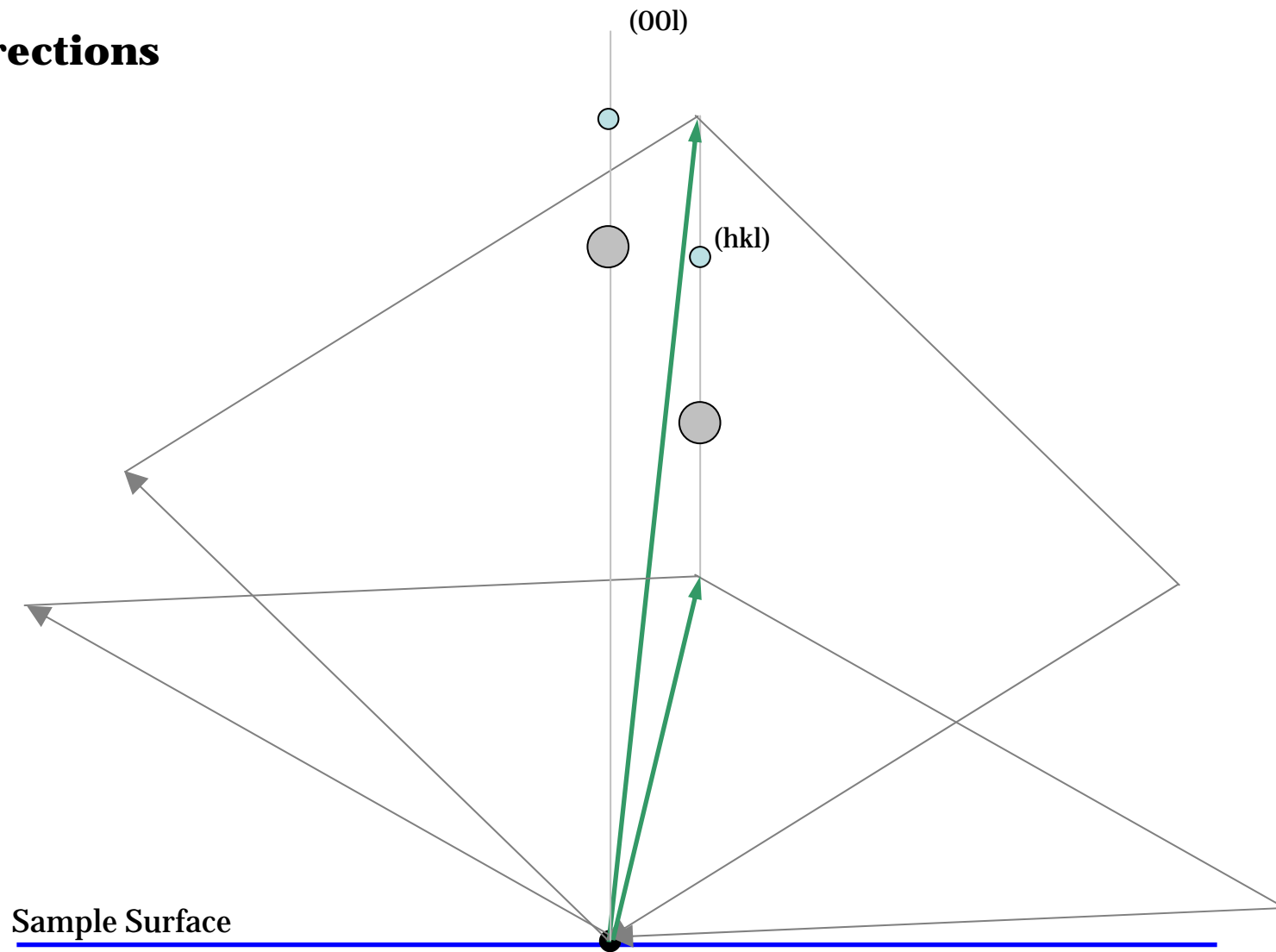


Strained Layer

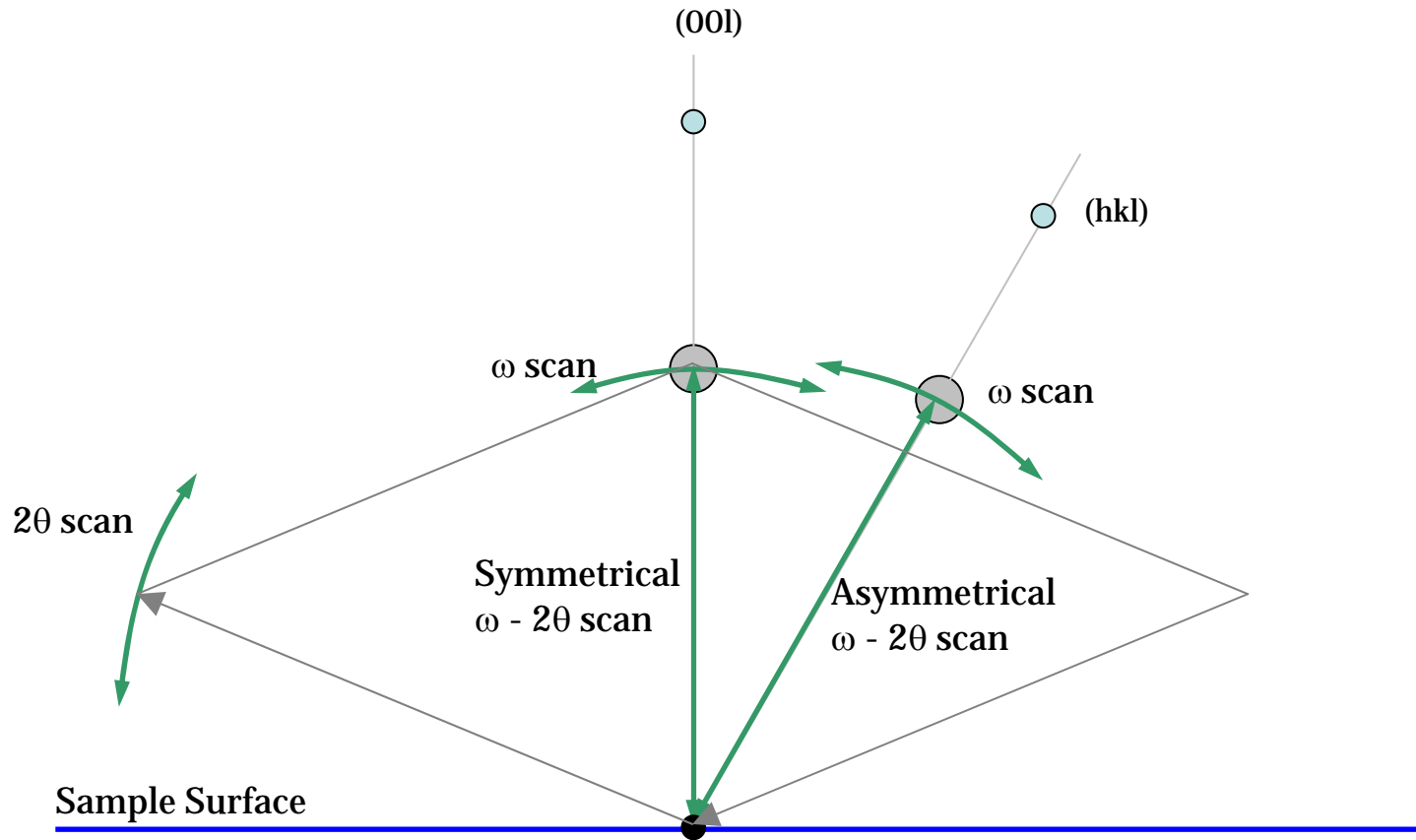
# Scan Directions



# Scan Directions



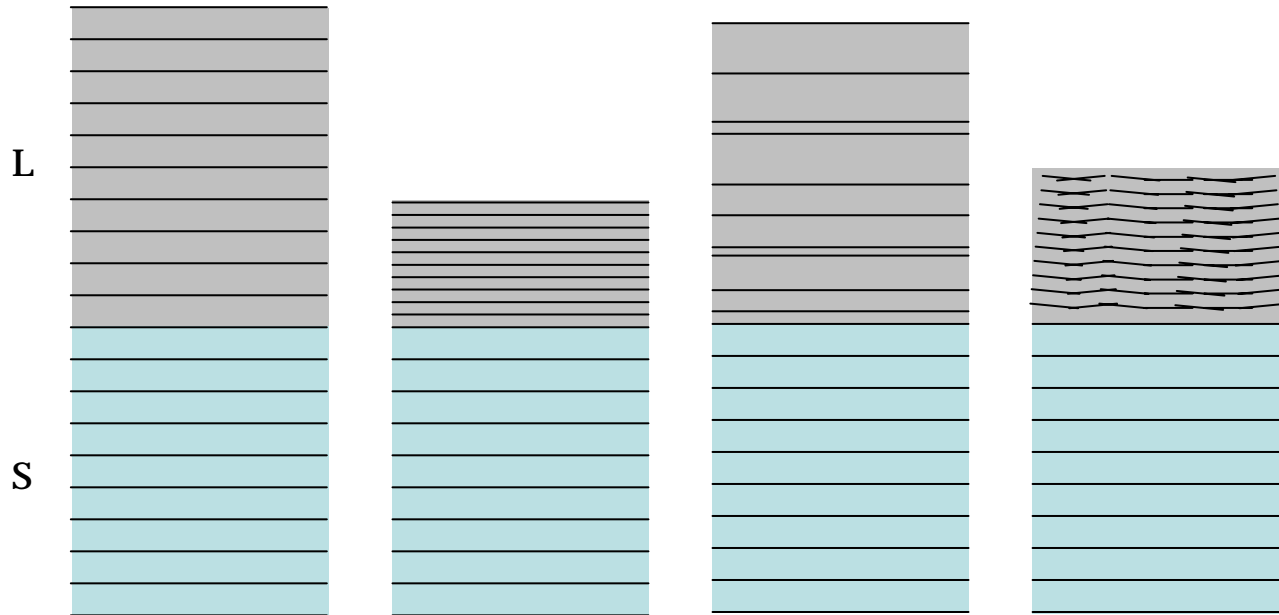
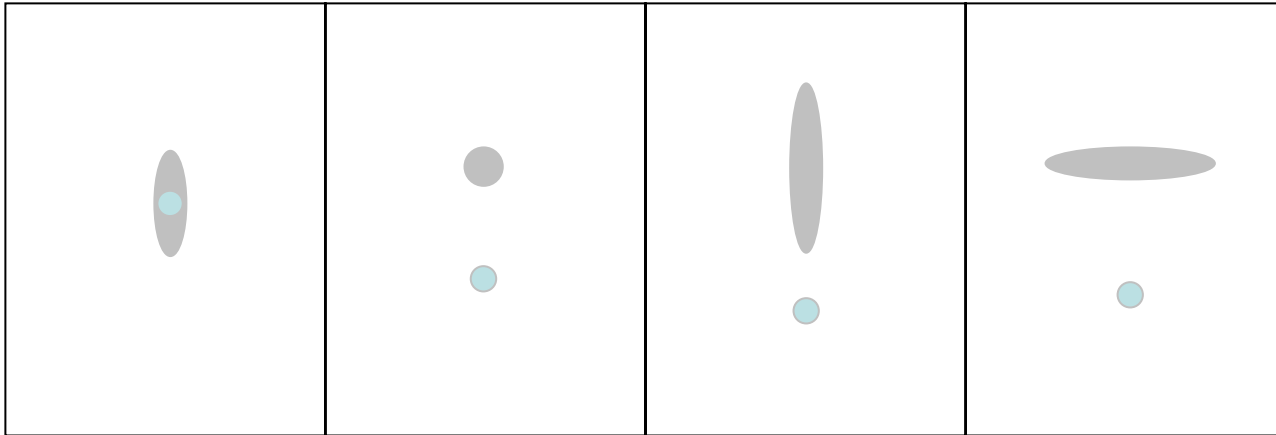
# Scan directions



# Real RLP shapes

Finite thickness effect

$$c_L < a_S$$



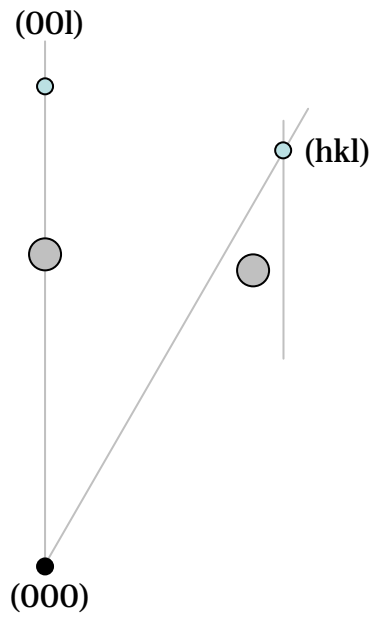
Homoepitaxy

Heteroepitaxy  
Tensile stress

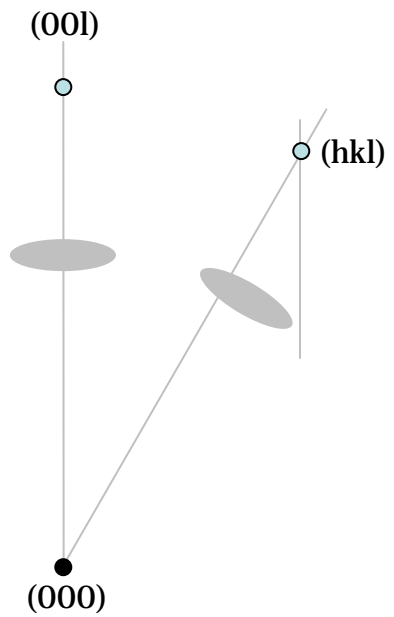
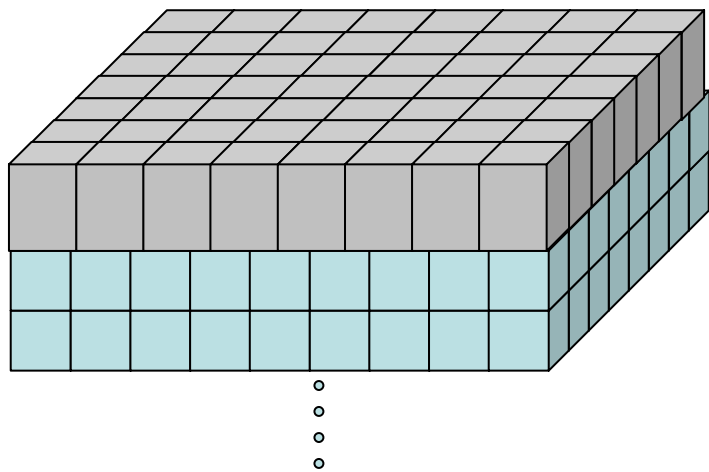
Heteroepitaxy  
d-spacing variation

Heteroepitaxy  
Mosaicity

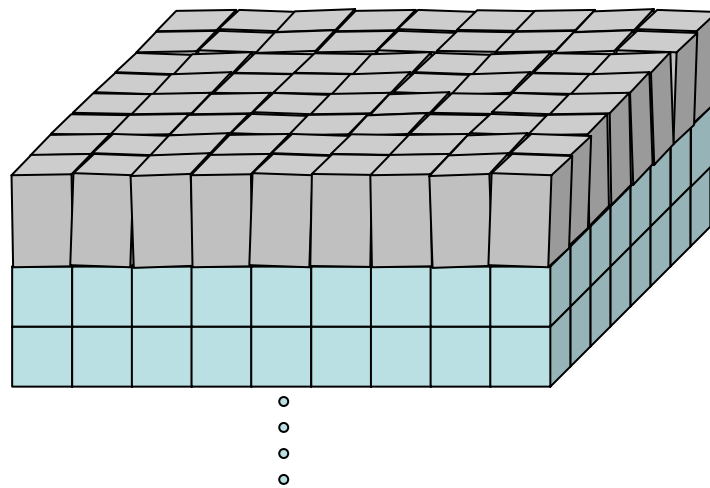




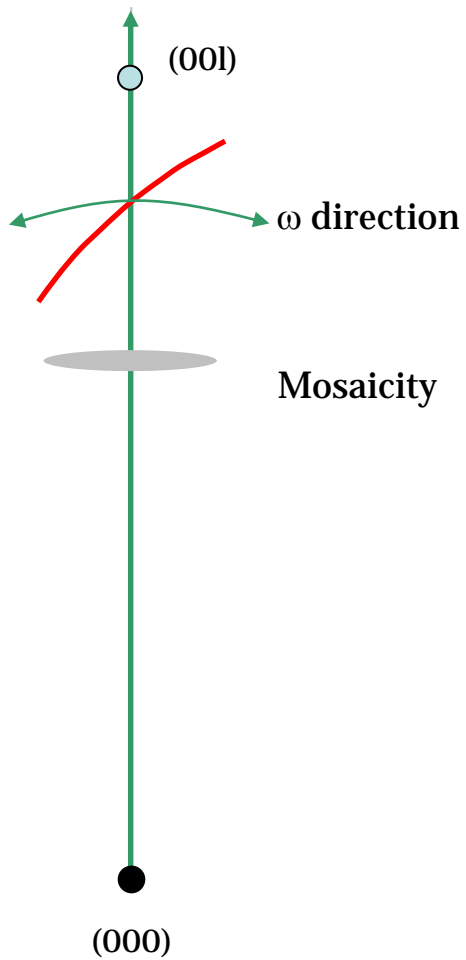
Partially Relaxed



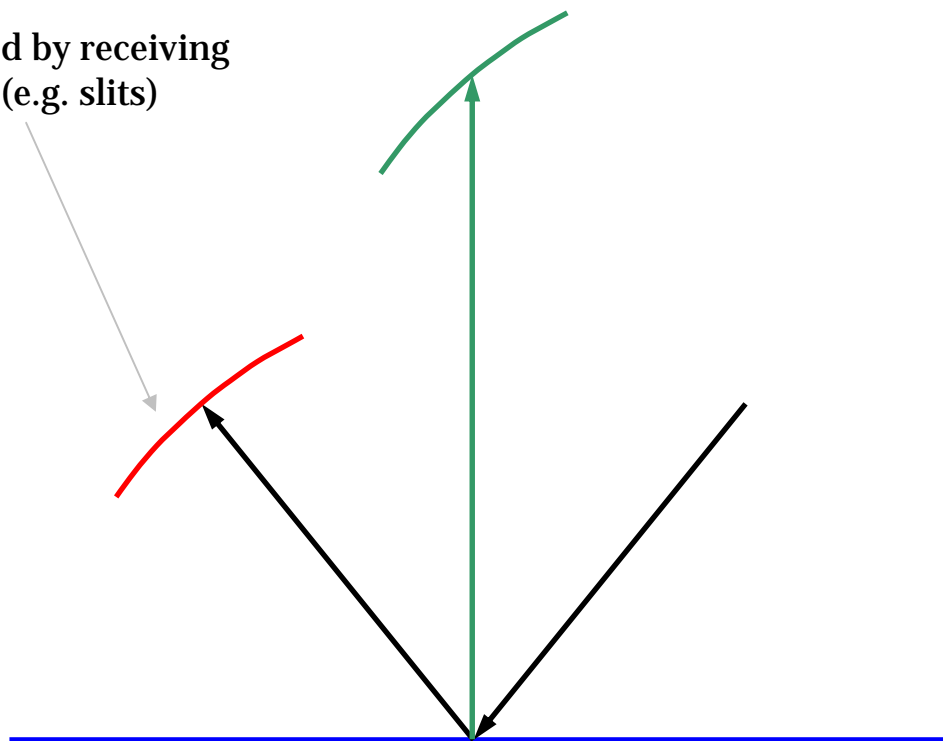
Partially  
Relaxed + Mosaicity



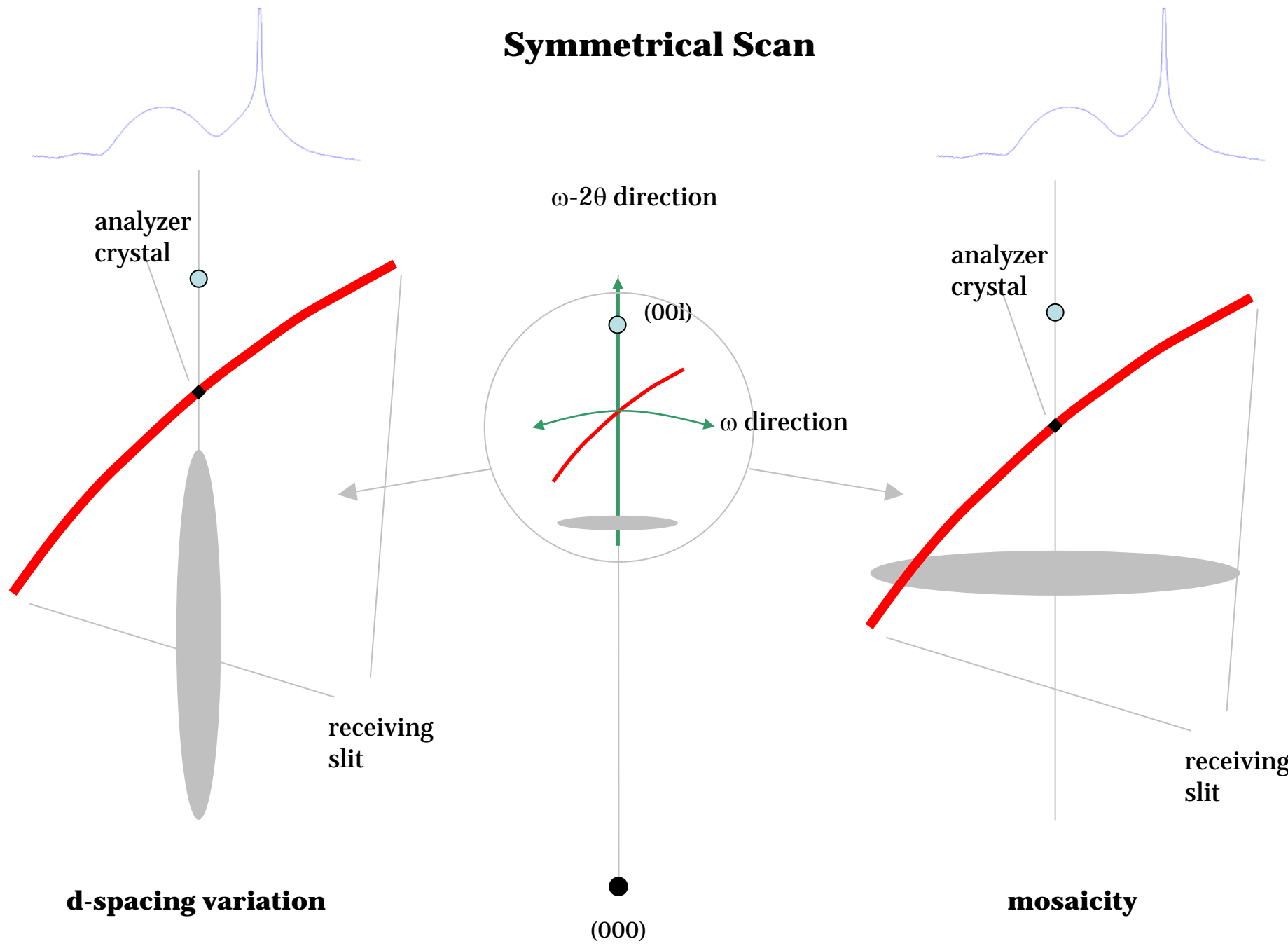
$\omega$ - $2\theta$  direction



Defined by receiving optics (e.g. slits)



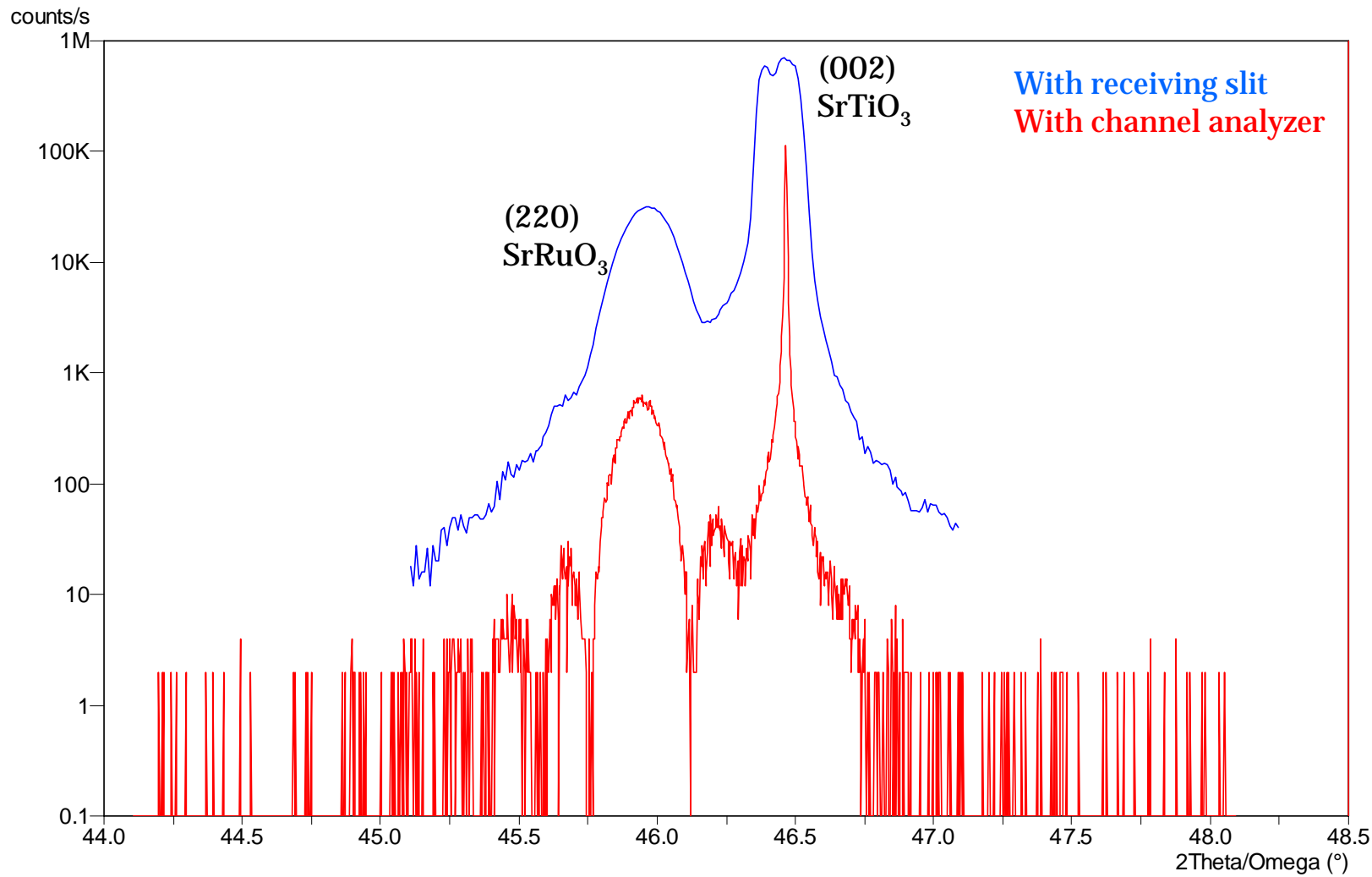
# Symmetrical Scan



**d-spacing variation**

(000)

**mosaicity**



# Mismatch

True lattice mismatch is:

$$m = \frac{a_L - a_S}{a_S}$$

The peak separation between substrate and layer is related to the change of interplanar spacing normal to the substrate through the equation:

$$\frac{\delta d}{d} = -\delta\theta \cot \theta$$

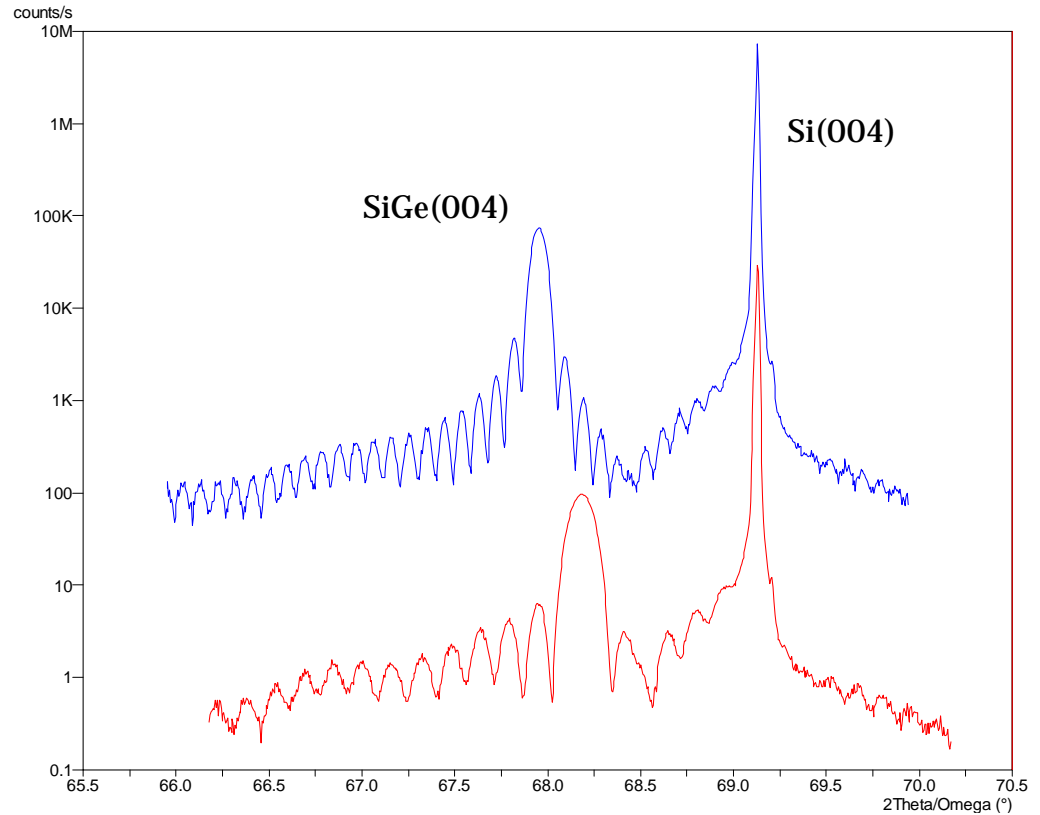
If it is (00l) reflection then the “experimental x-ray mismatch”:

$$m^* = \frac{\delta a}{a} = \frac{\delta d}{d}$$

And true mismatch can be obtained through:

$$m = m^* \left\{ \frac{1-\nu}{1+\nu} \right\}$$

where:  $\nu$  – Poisson ratio



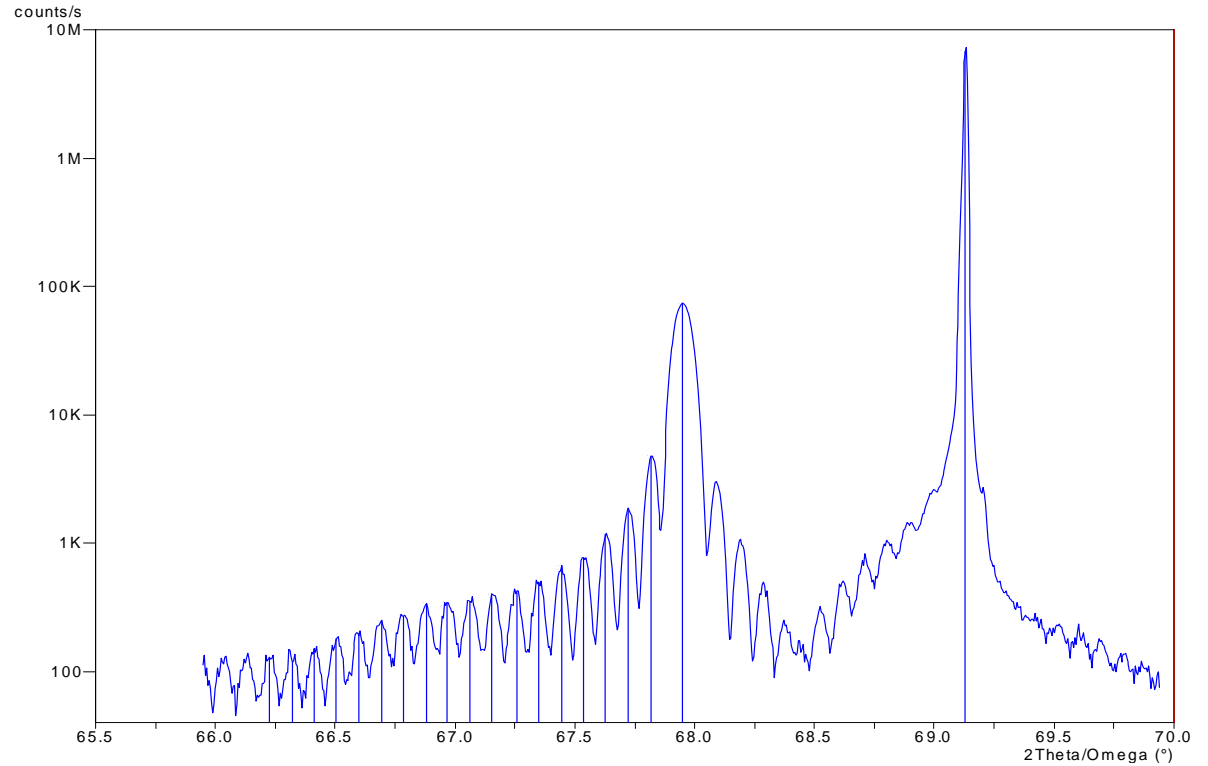
$$\nu \approx \frac{1}{3}$$

$$m \approx \frac{m^*}{2}$$

# Layer Thickness

Interference fringes observed in the scattering pattern, due to different optical paths of the x-rays, are related to the thickness of the layers

$$t = \frac{(n_1 - n_2)\lambda}{2(\sin \omega_1 - \sin \omega_2)}$$



## Substrate Layer Separation

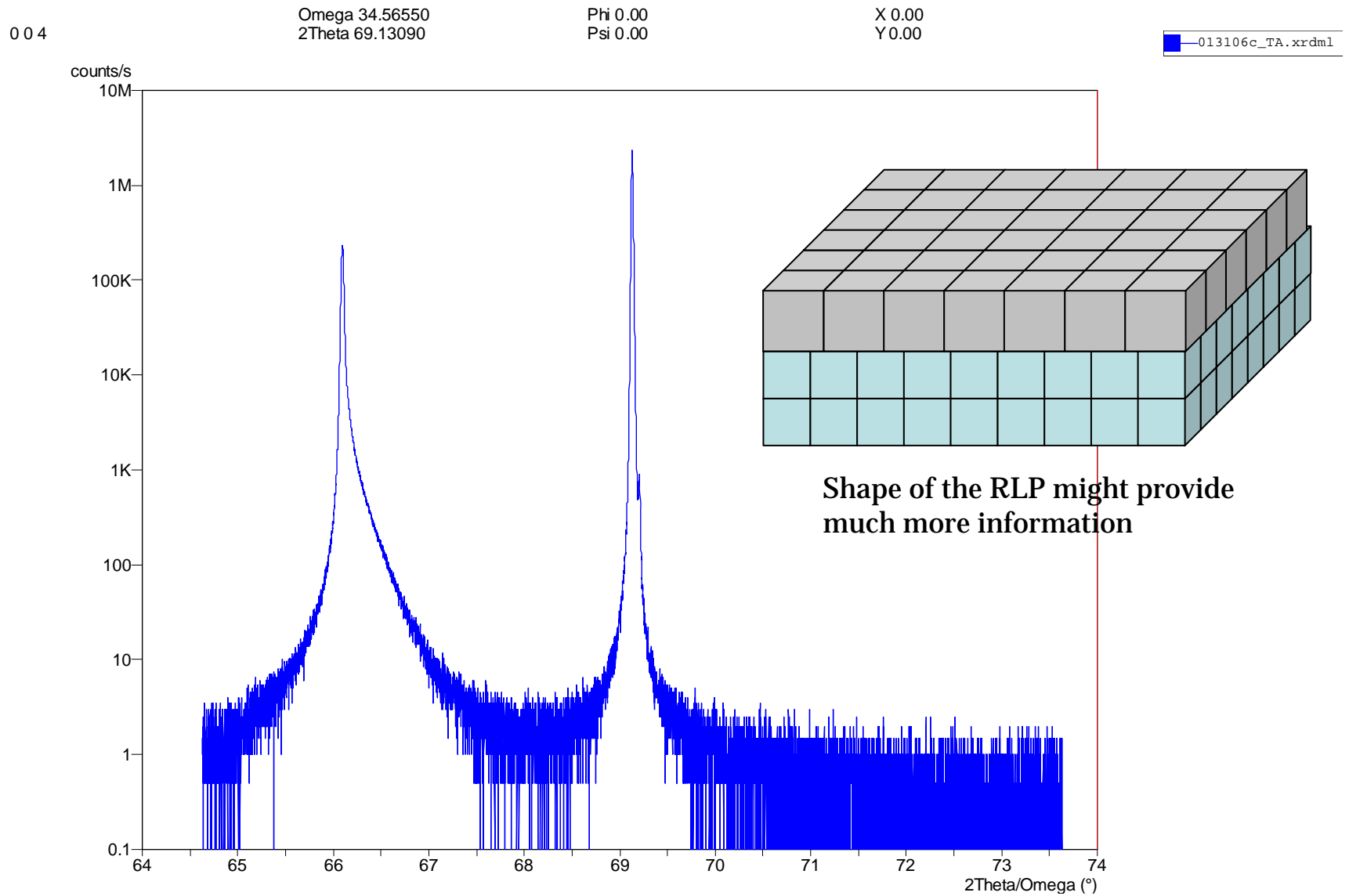
S-peak:		L-peak:		Separation:	
Omega(°)	34.5649	Omega(°)	33.9748	Omega(°)	0.59017
2Theta(°)	69.1298	2Theta(°)	67.9495	2Theta(°)	1.18034

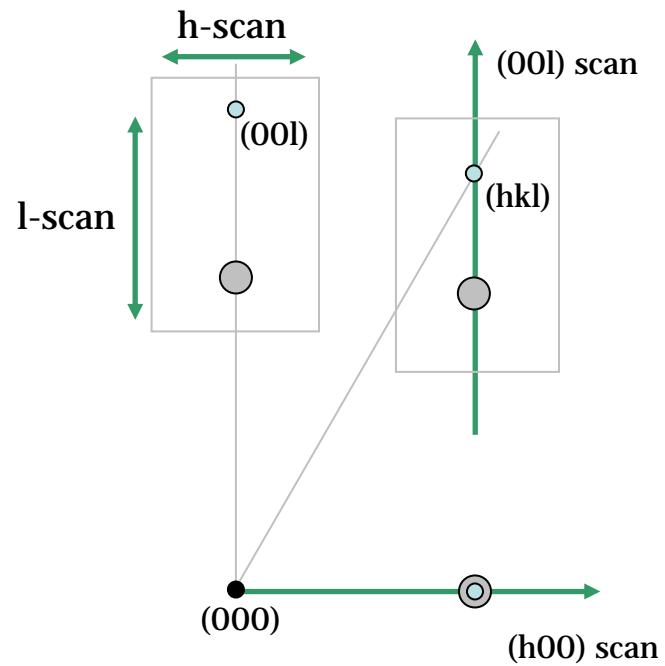
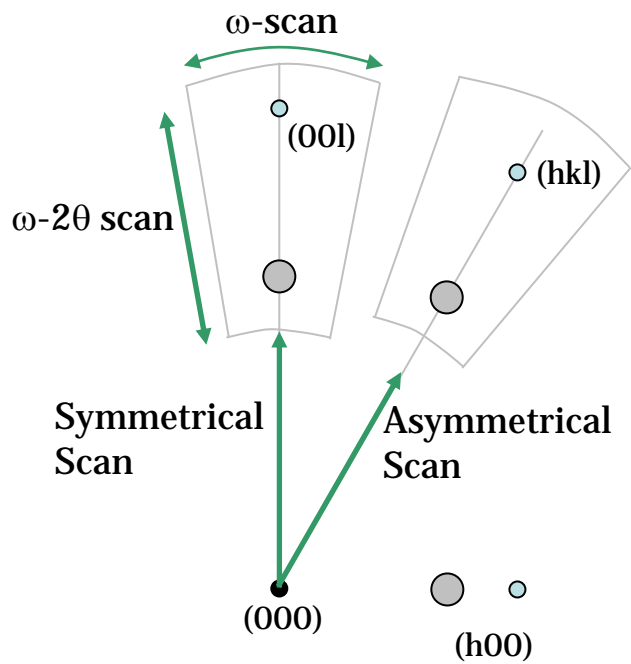
## Layer Thickness

Mean fringe period (°): 0.09368  
 Mean thickness (um): 0.113 ± 0.003

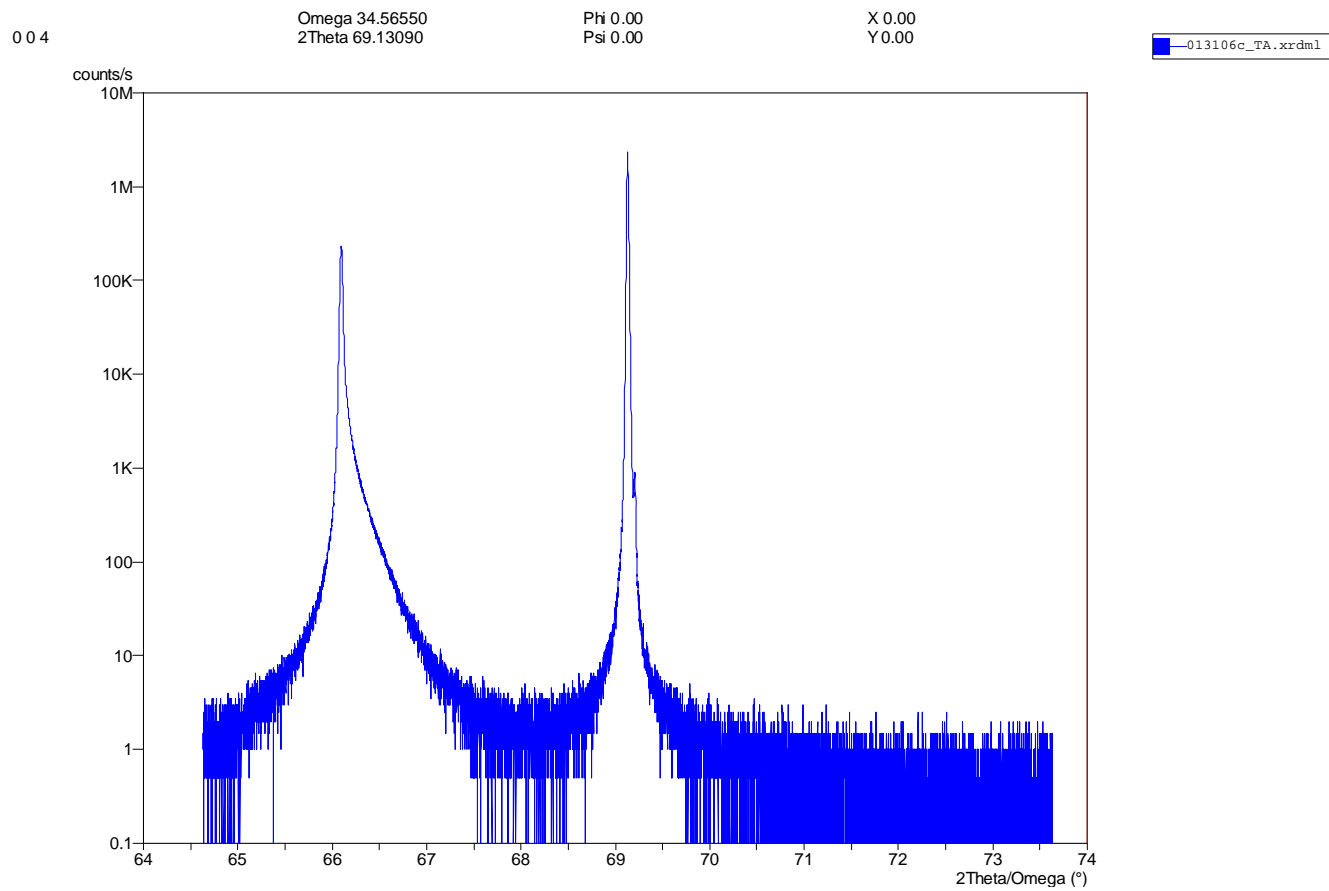
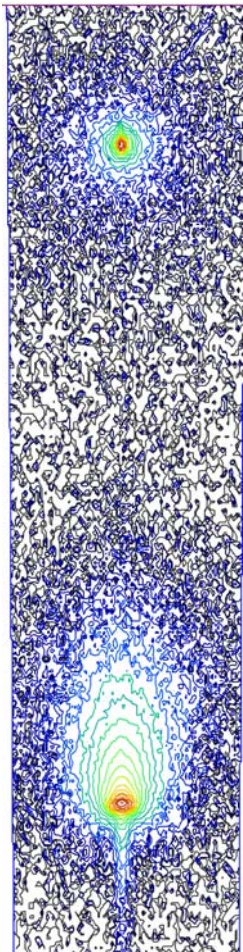
2Theta/Omega (°)	Fringe Period (°)	Thickness (um)
66.22698 - 66.32140	0.09442	0.111637
66.32140 - 66.41430	0.09290	0.113528
66.41430 - 66.50568	0.09138	0.115481
66.50568 - 66.59858	0.09290	0.113648
66.59858 - 66.69300	0.09442	0.111878
66.69300 - 66.78327	0.09027	0.117079

# Relaxed SiGe on Si(001)





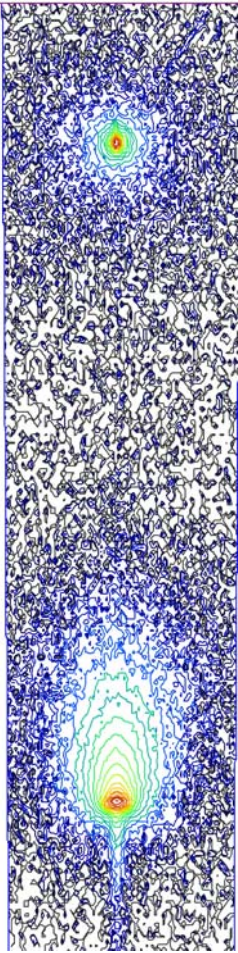




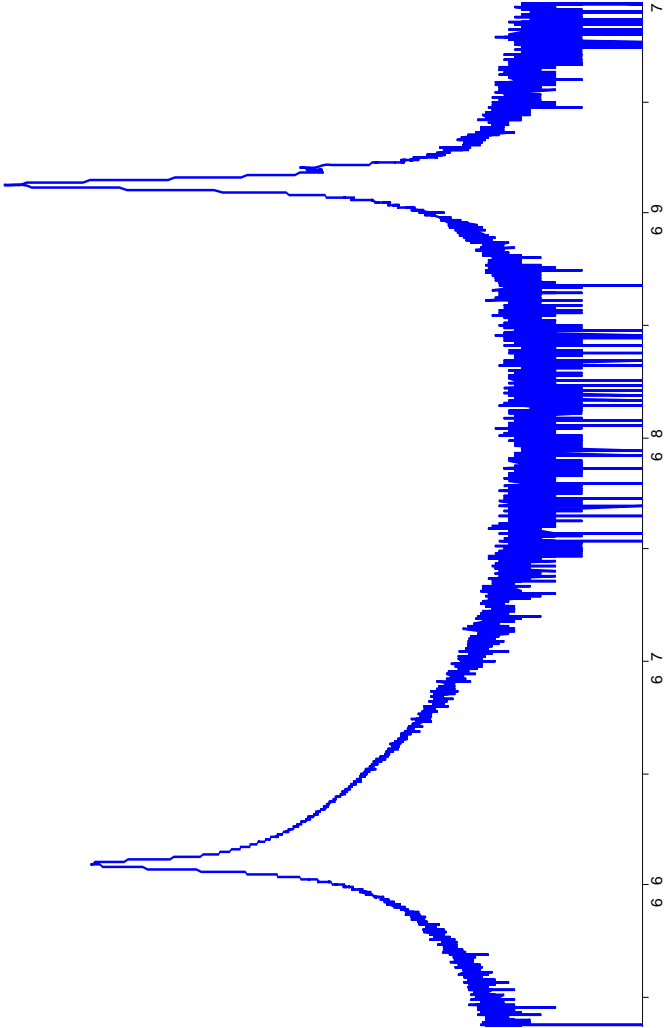
# Relaxed SiGe on Si(001)

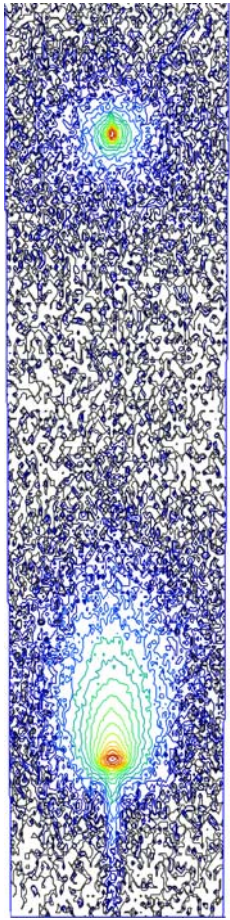
(004) RLM

Si(004)

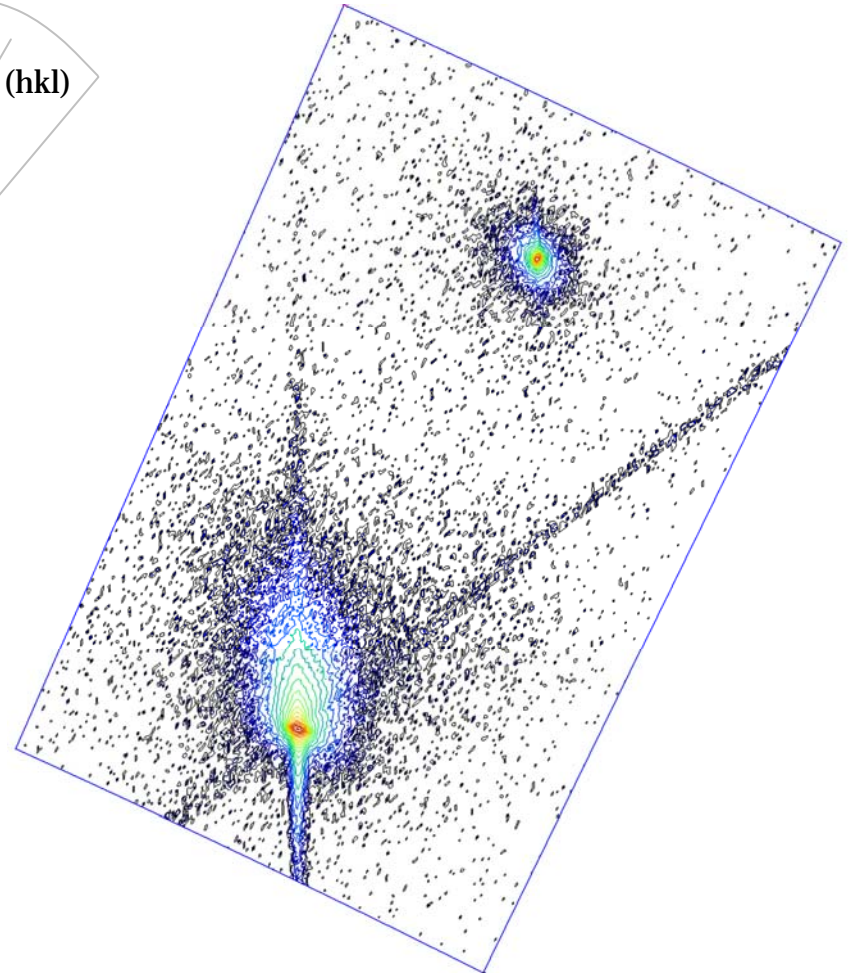
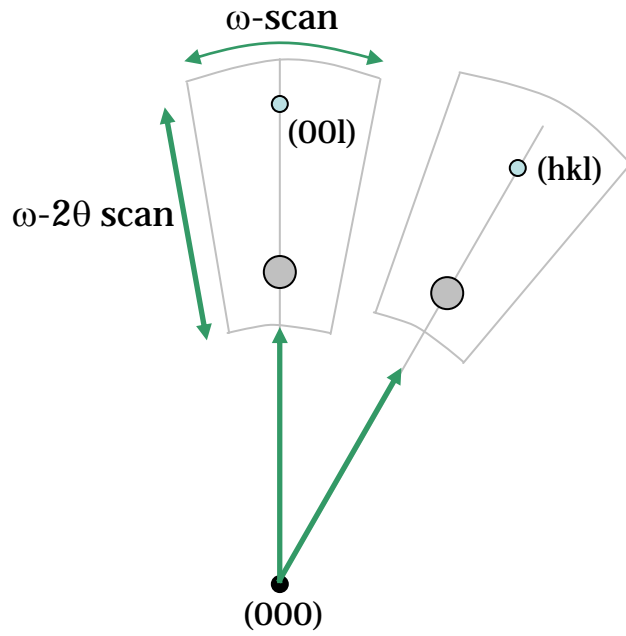


SiGe(004)





(004)



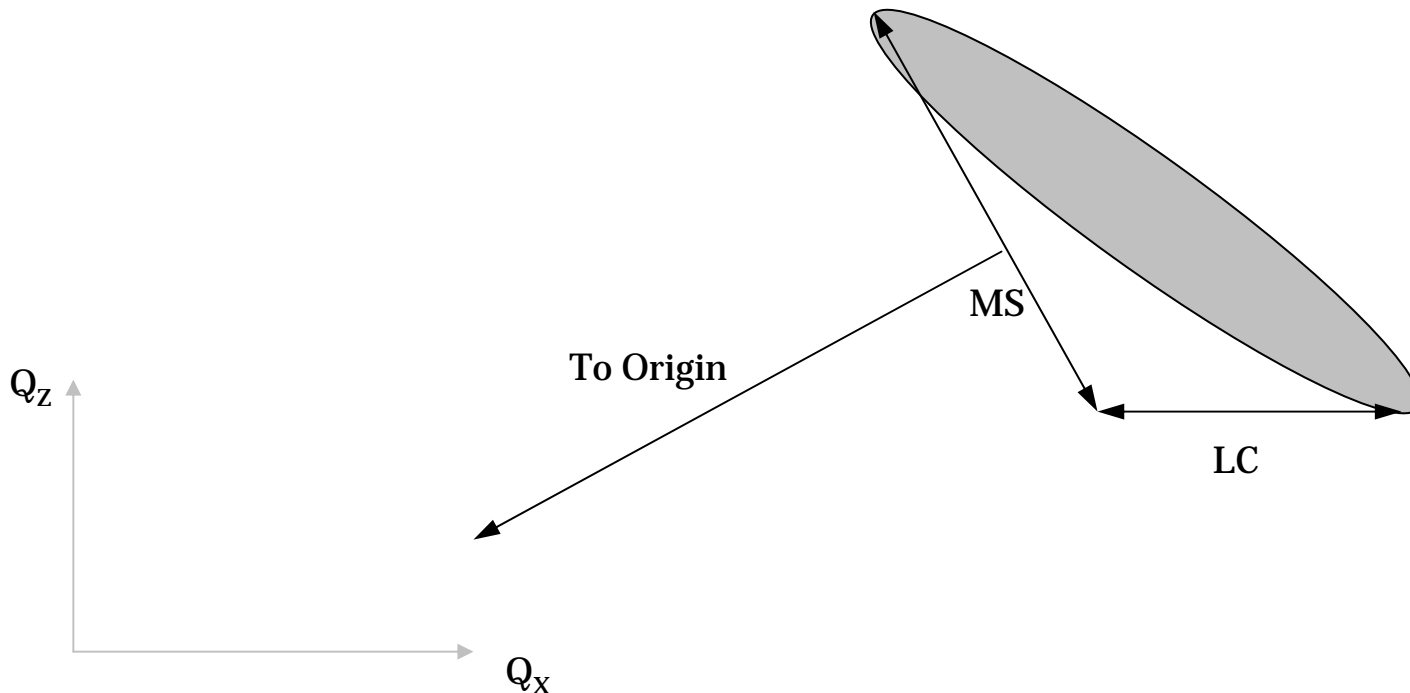
(113)

## Mosaic Spread and Lateral Correlation Length

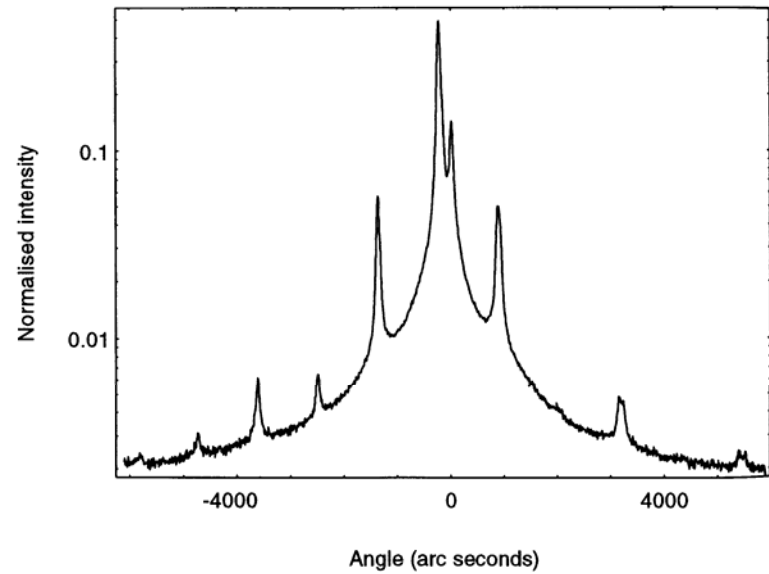
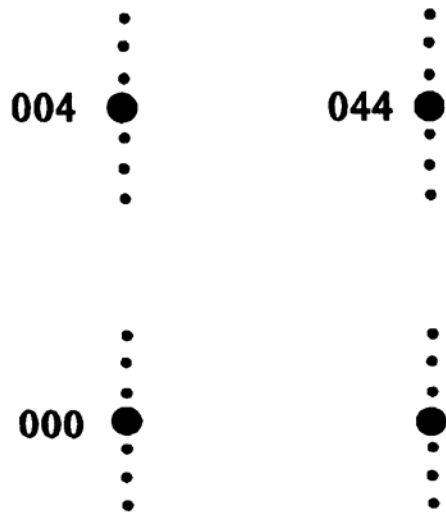
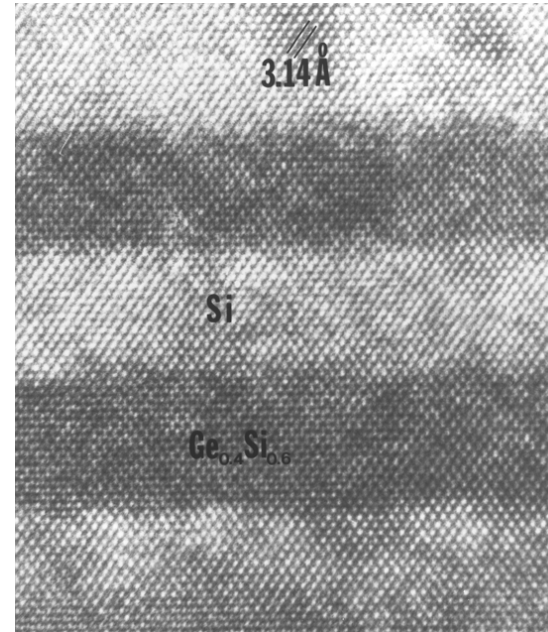
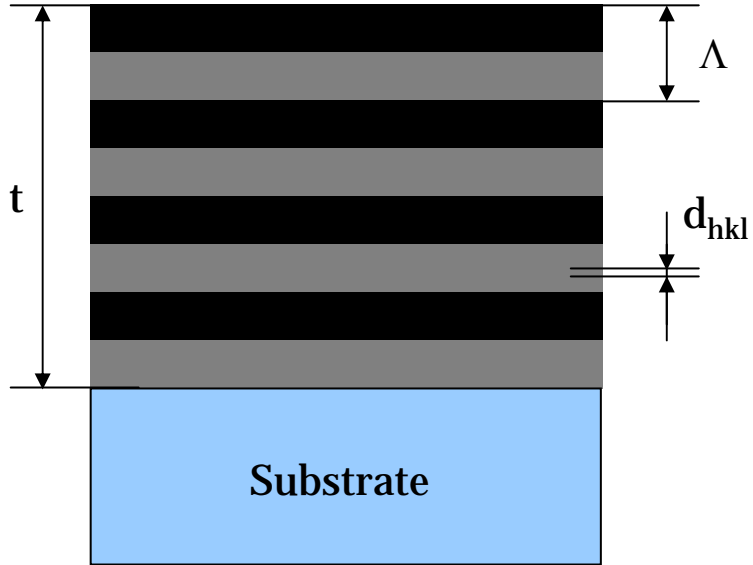
The Mosaic Spread and Lateral Correlation Length functionality derives information from the shape of a layer peak in a diffraction space map recorded using an asymmetrical reflection

The mosaic spread of the layer is calculated from the angle that the layer peak subtends at the origin of reciprocal space measured perpendicular to the reflecting plane normal.

The lateral correlation length of the layer is calculated from the reciprocal of the FWHM of the peak measured parallel to the interface.



# Superlattices and Multilayers



# Superlattices and Multilayers

(001)



(000)

2



(001)



(000)

4



(001)



(000)

6



(001)



(000)

10



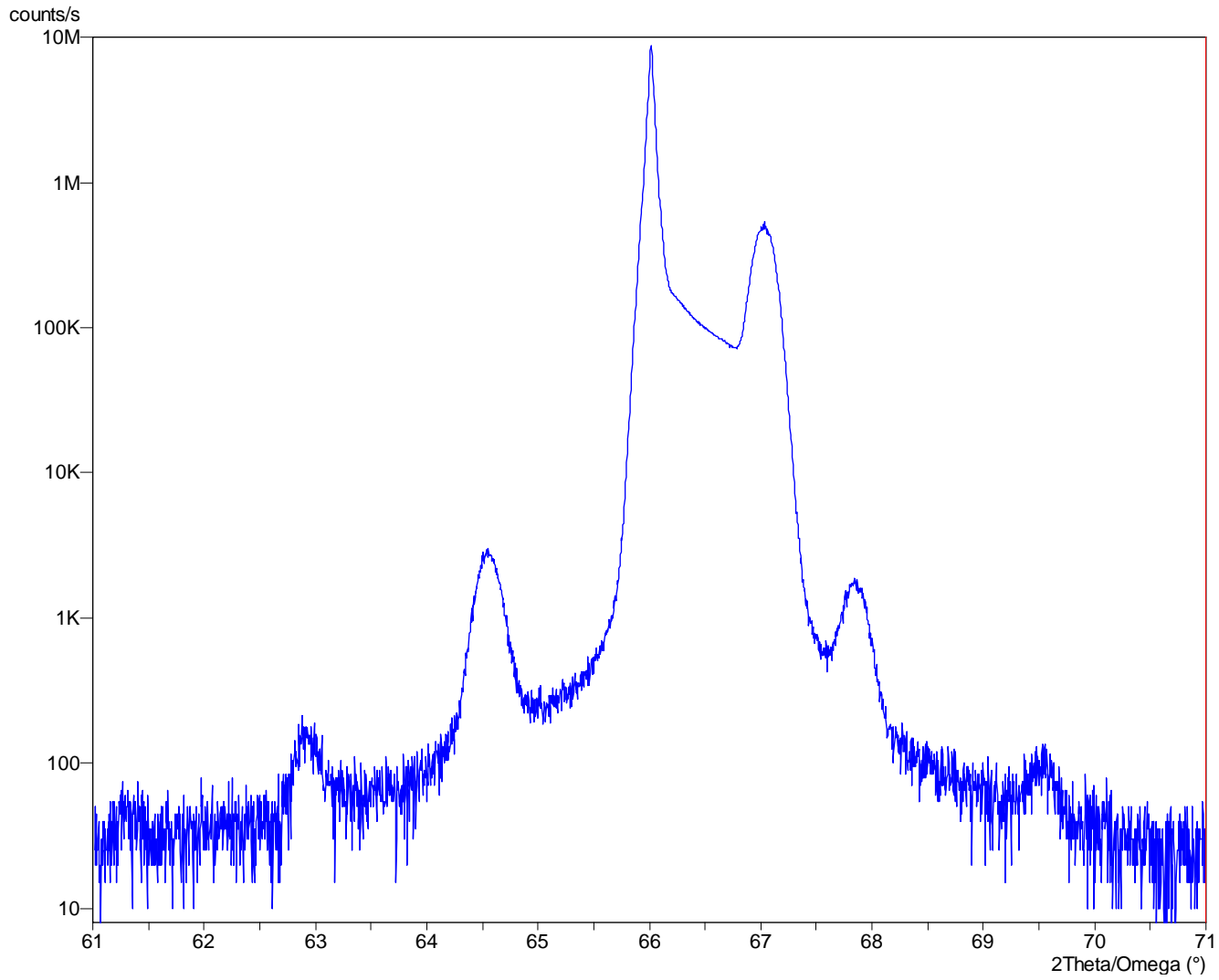
0 0 4

Omega 33.00650  
2Theta 66.01310

Phi 0.00  
Psi 0.00

X 0.00  
Y 0.00

3683ss1.xrdml





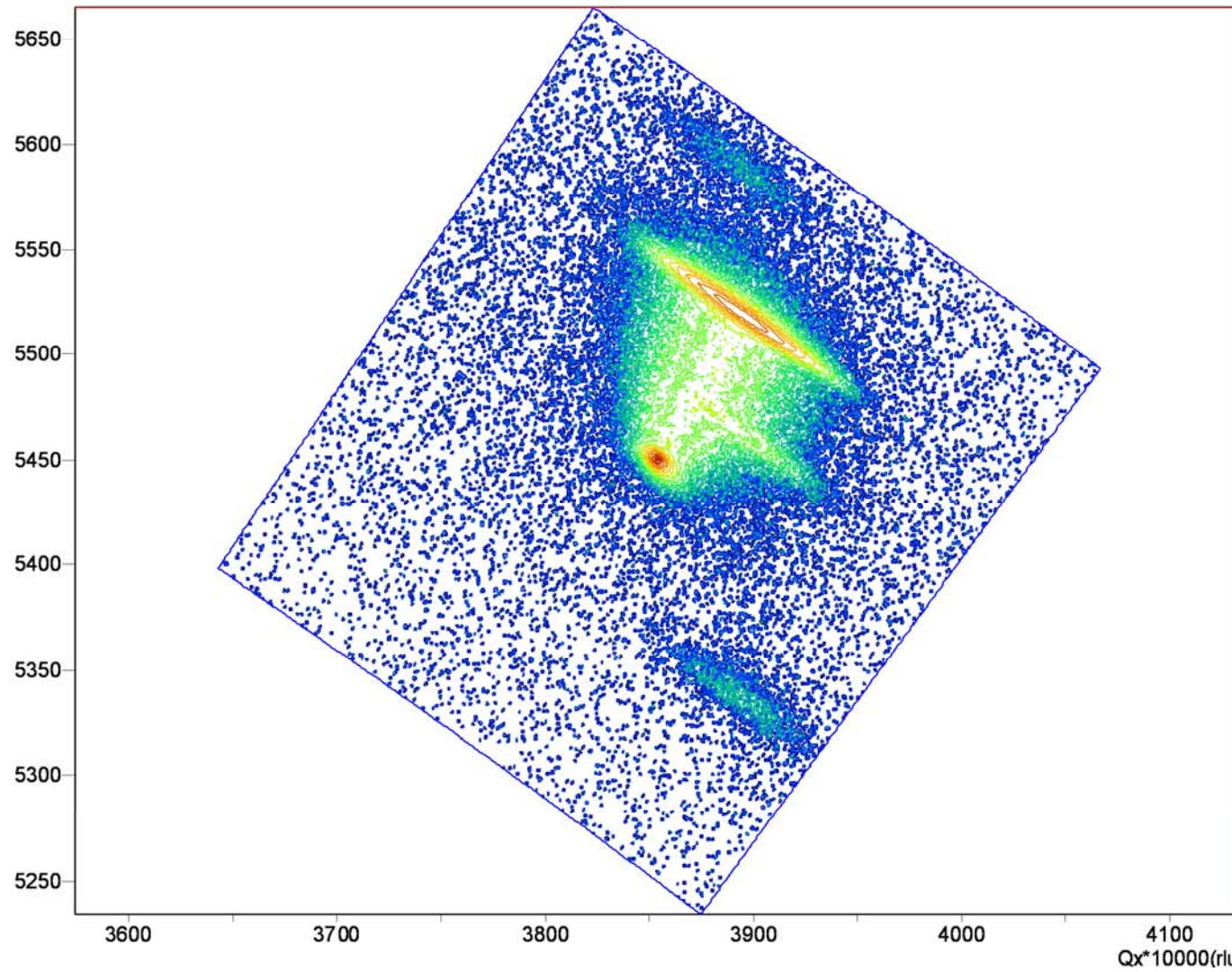
Omega 6.61060  
2Theta 83.75000

Phi 0.00  
Psi 0.00

X 0.00  
Y 13.00  
Z 9.110

SLAC\_671.xrdml

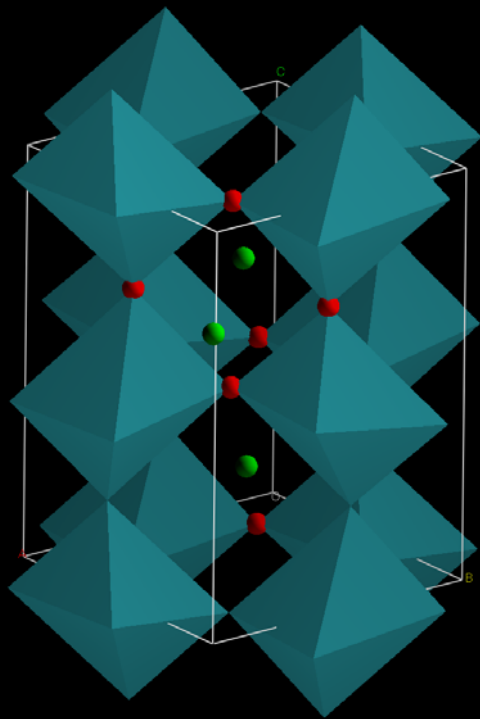
Qy\*10000(rlu)





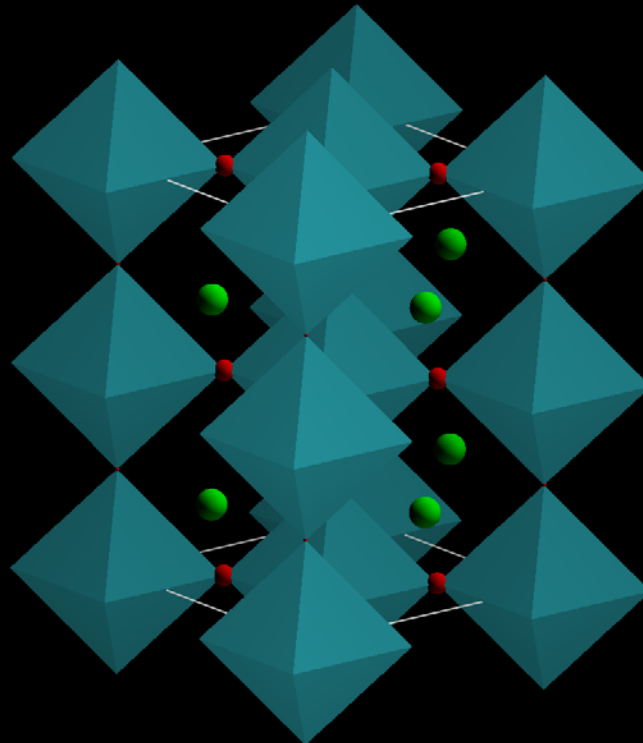
# Structure of SrRuO<sub>3</sub>

Orthorhombic



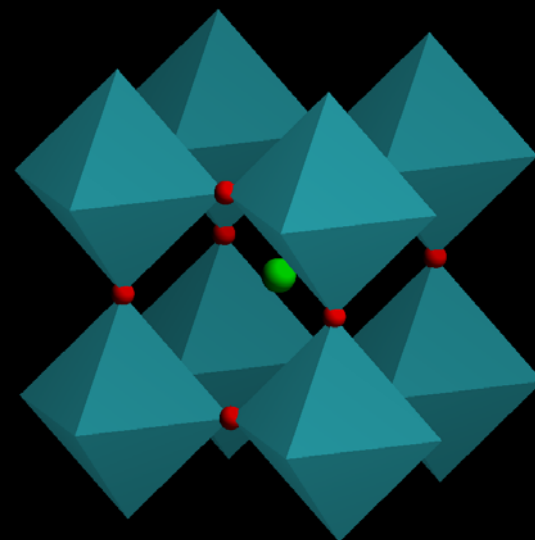
$a = 5.586 \text{ \AA}$   
 $b = 5.555 \text{ \AA}$   
 $c = 7.865 \text{ \AA}$

Tetragonal



$a = 5.578 \text{ \AA}$   
 $c = 7.908 \text{ \AA}$

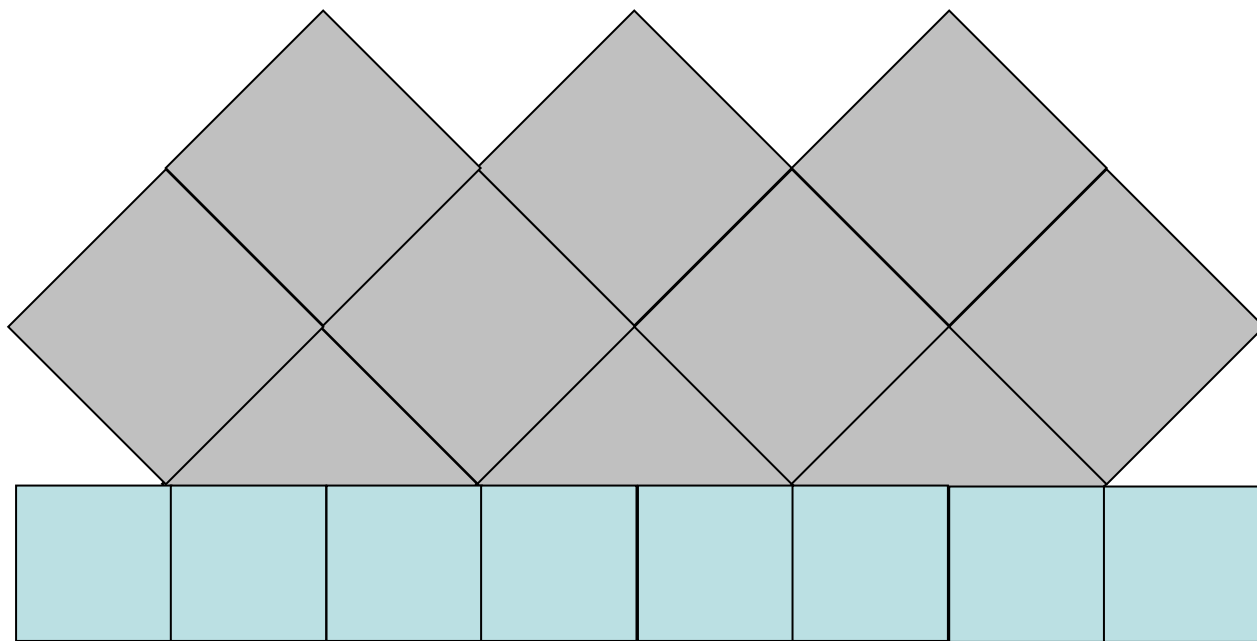
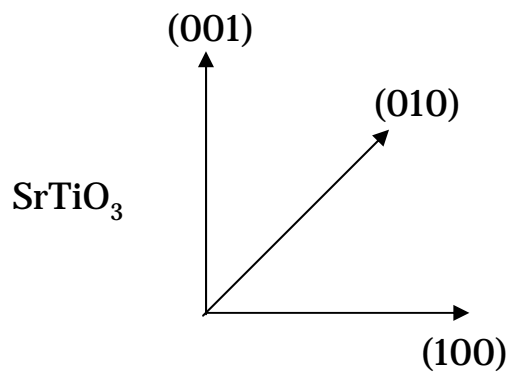
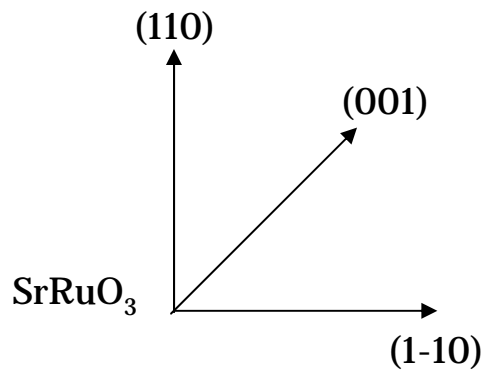
Cubic



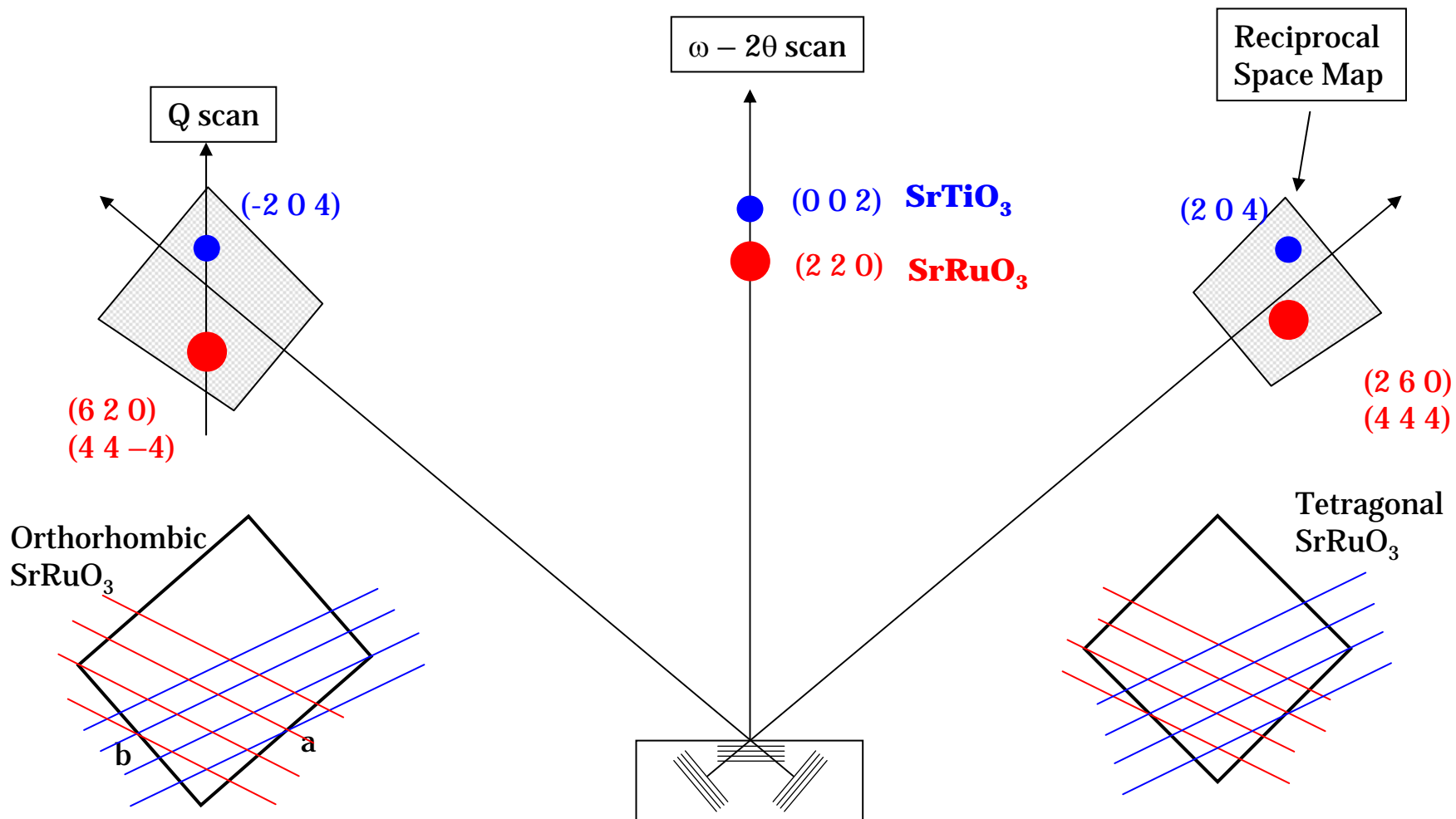
$a = 3.956 \text{ \AA}$

→  
 $275\text{-}550 \text{ }^\circ\text{C}$

→  
 $510\text{-}702 \text{ }^\circ\text{C}$

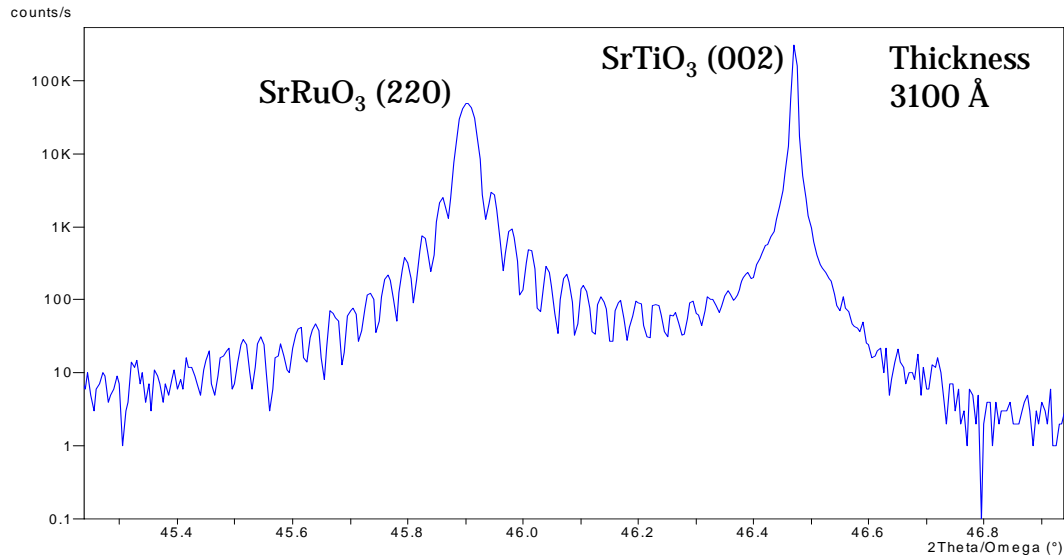
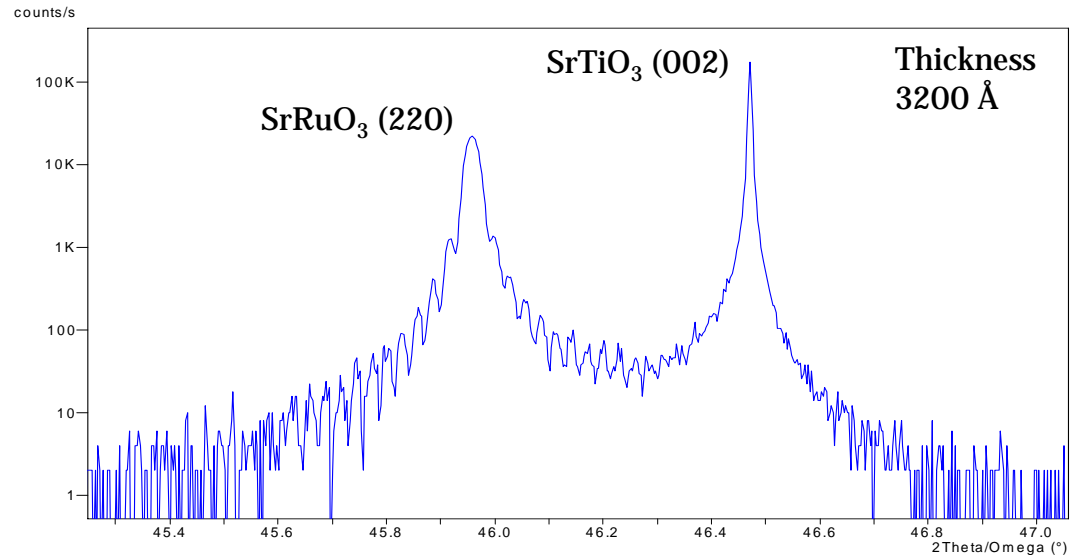


# X-ray Diffraction Scan Types



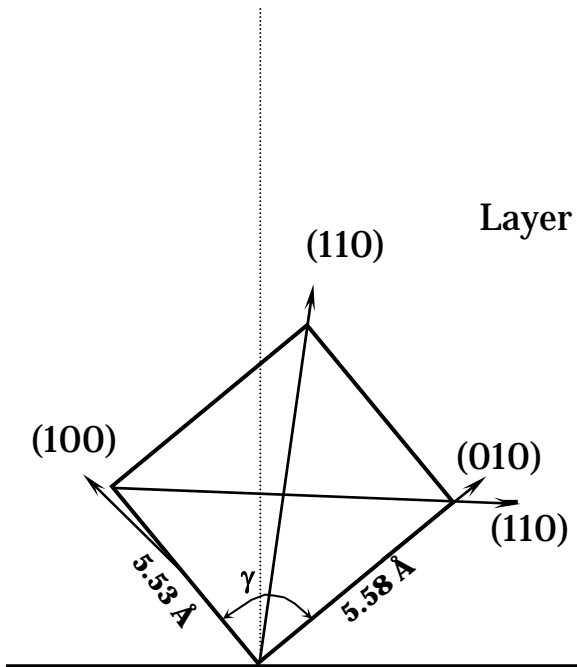
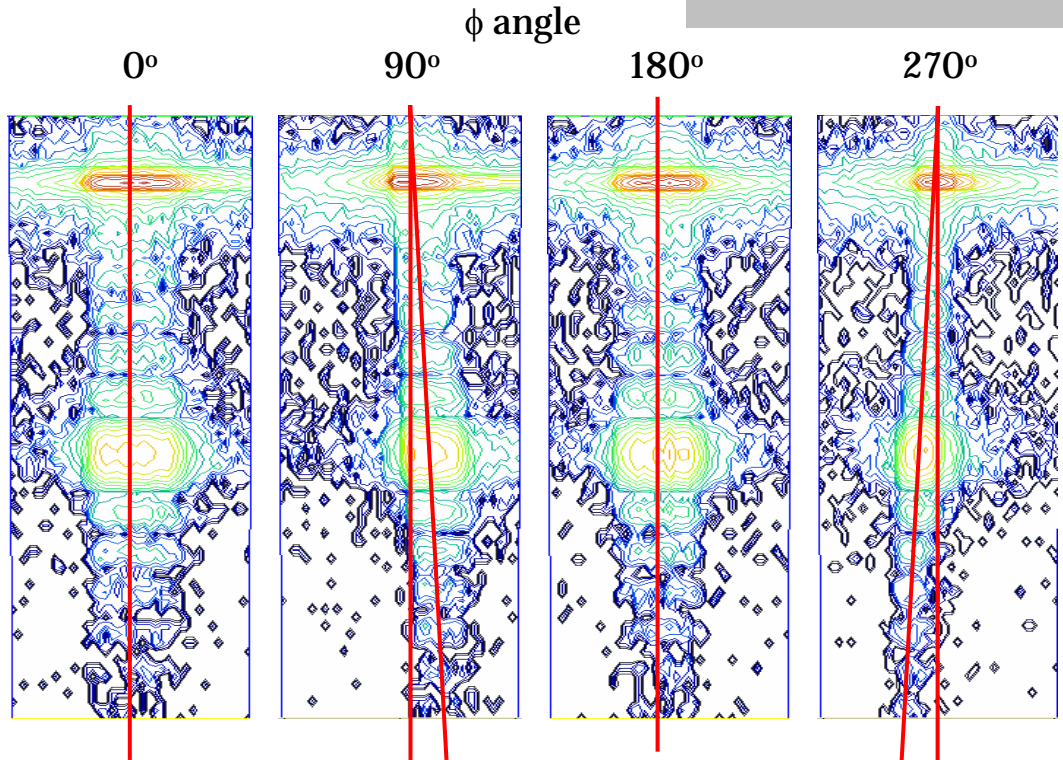
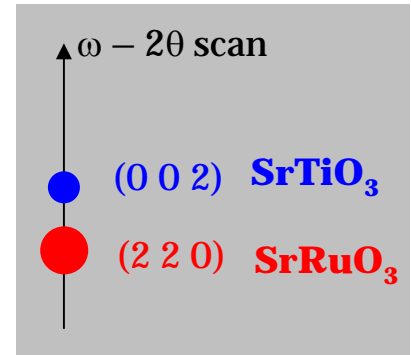
# $\omega - 2\theta$ symmetrical scans

Finite size fringes indicate well ordered films

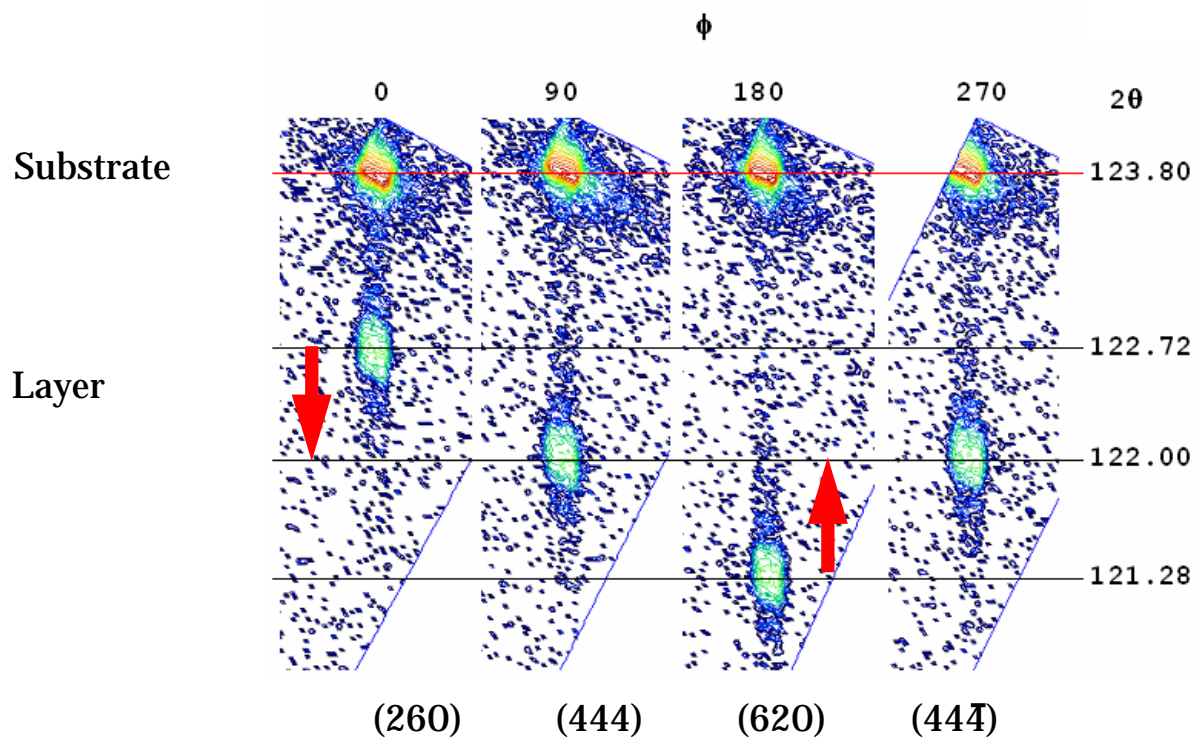
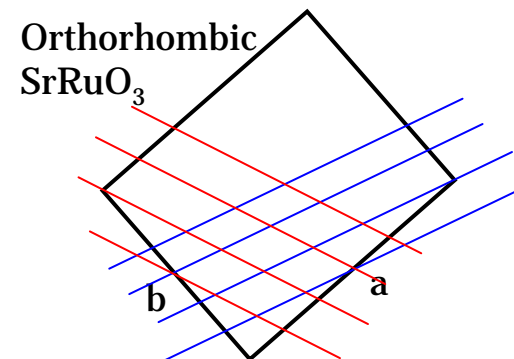


# Reciprocal Lattice Map of SrRuO<sub>3</sub> (220) and SrTiO<sub>3</sub> (002)

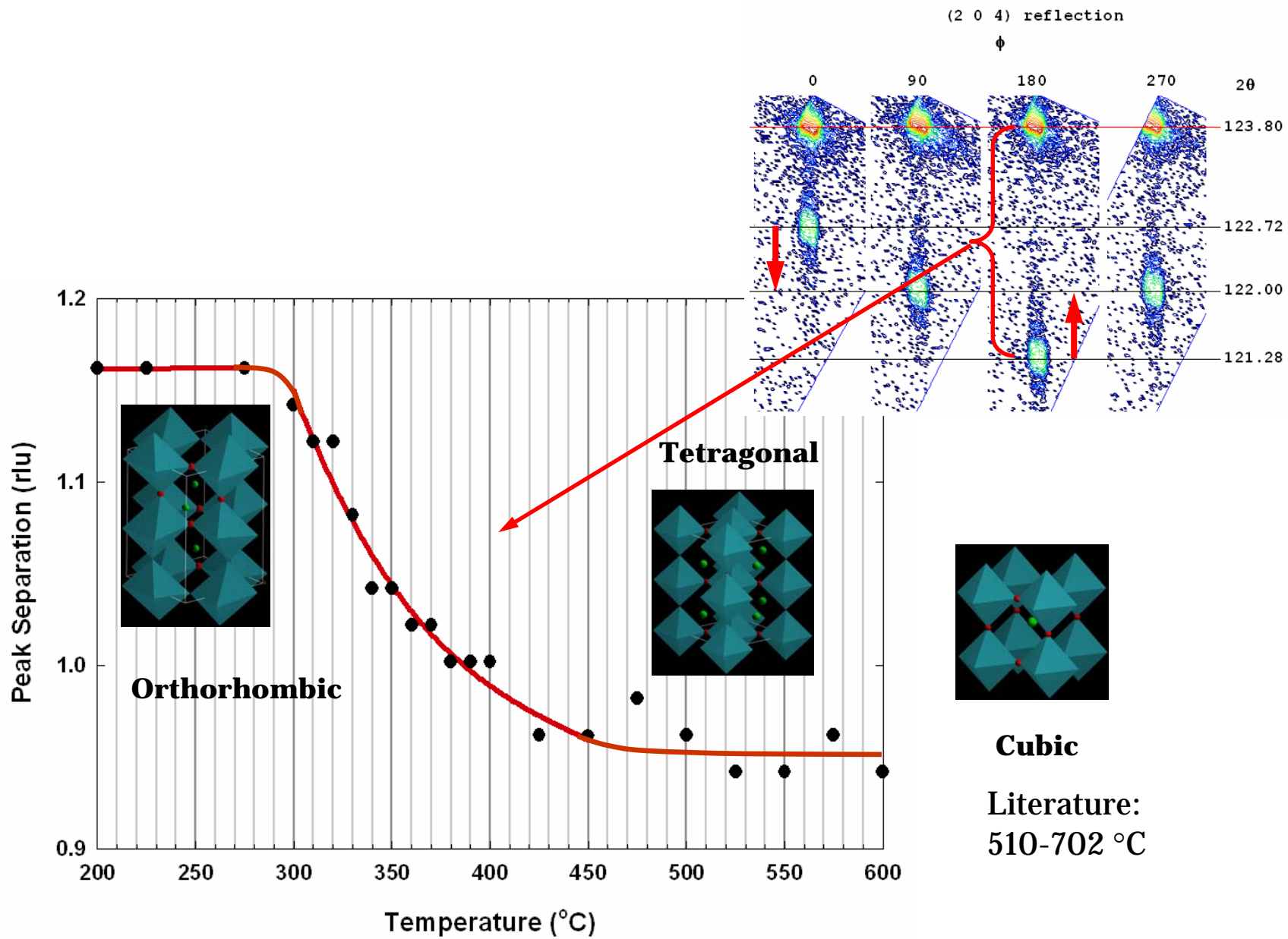
Distorted perovskite structure:  
Films are slightly distorted from orthorhombic,  $\gamma = 89.1^\circ - 89.4^\circ$



# High-Resolution Reciprocal Area Mapping



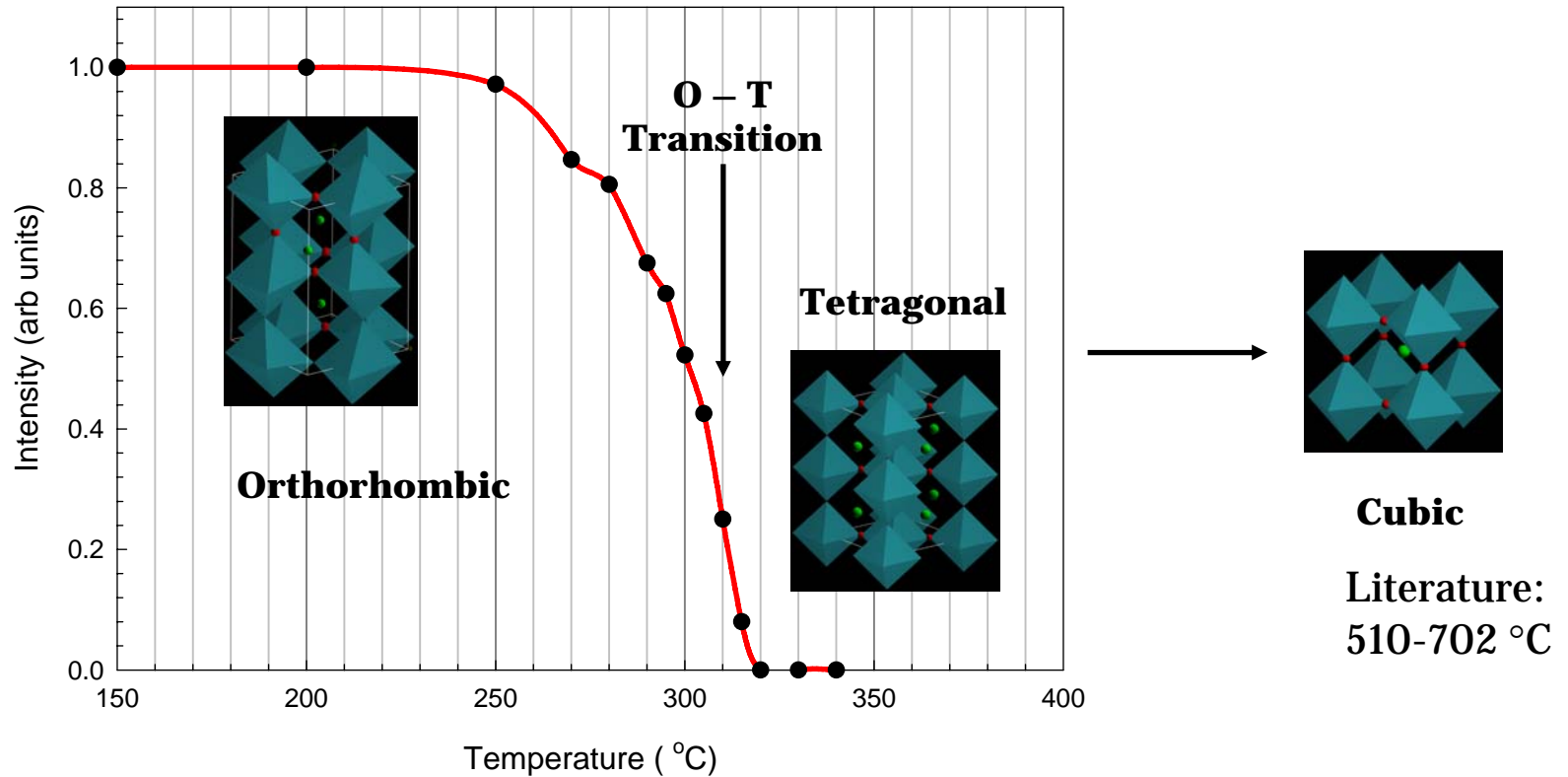
**Orthorhombic to Tetragonal Transition**



Transition Orthorhombic to Tetragonal  $\sim 350^{\circ}\text{C}$

# Structural Transition, (221) reflection

(221) Peak	
Orthorhombic	<b>Present</b>
Tetragonal	<b>Absent</b>

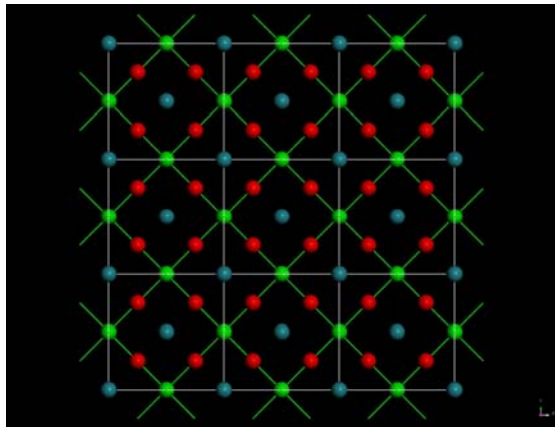
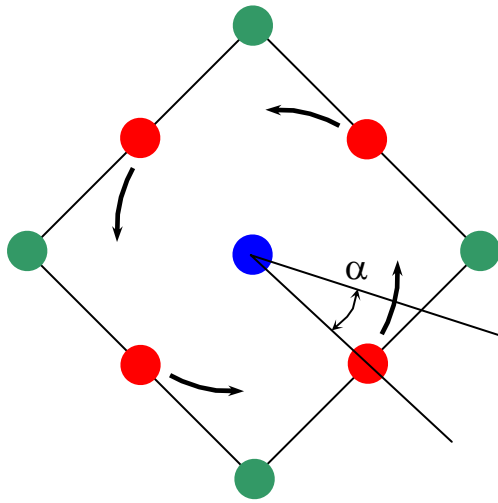


Transition Orthorhombic to Tetragonal ~ 310 °C

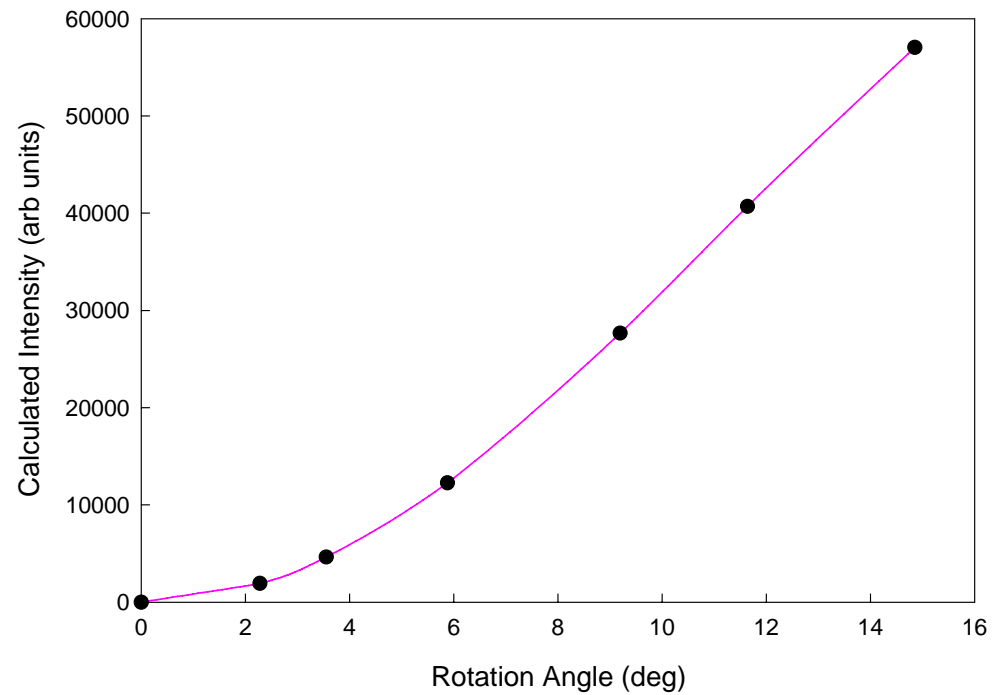
Transition Orthorhombic to Tetragonal ~ 310 °C



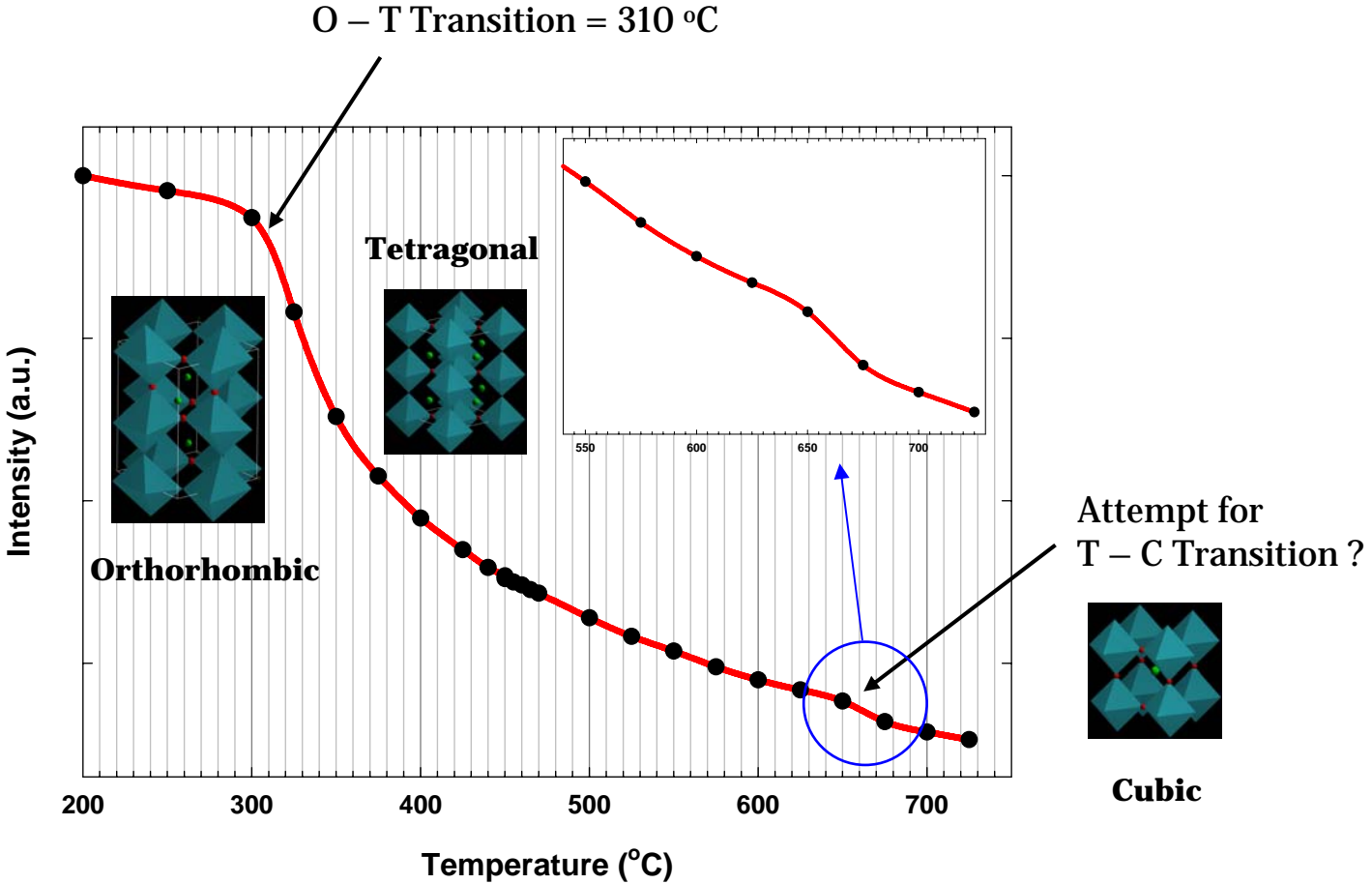
# Structural Transition, (211) reflection



(211) peak is **absent** in cubic  $\text{SrRuO}_3$

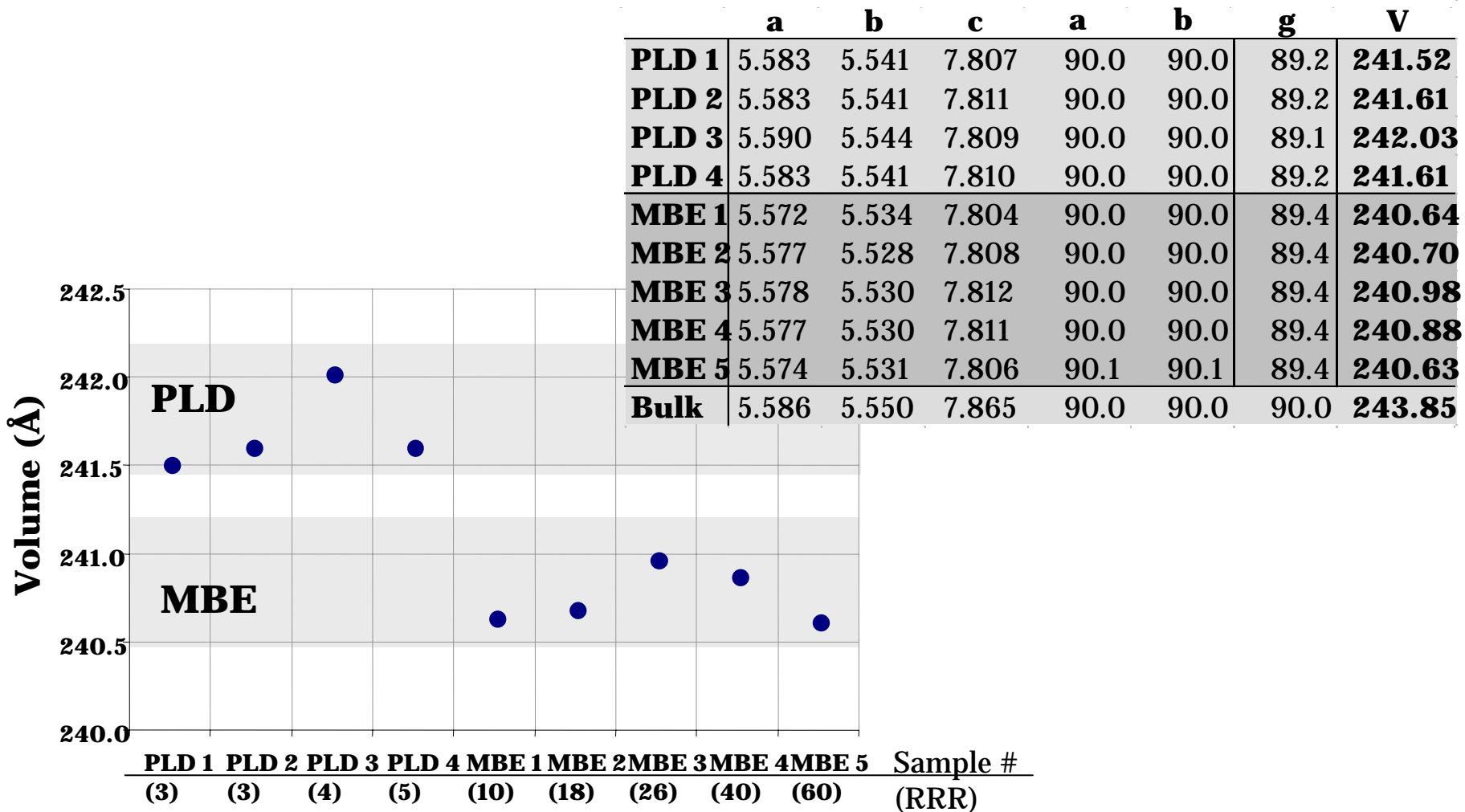


# Structural Transition, (211) reflection



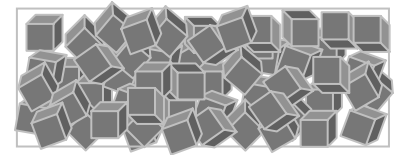
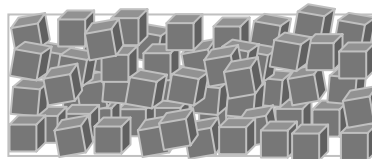
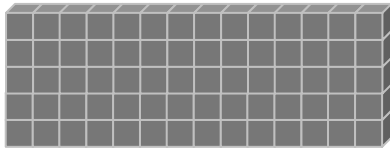
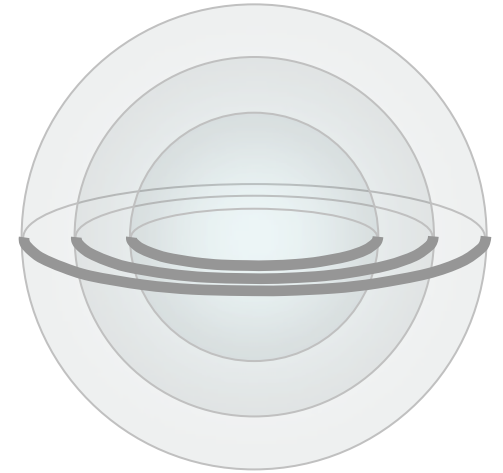
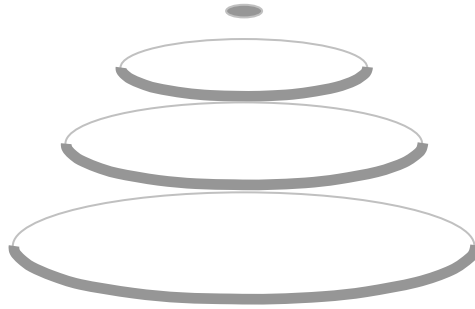
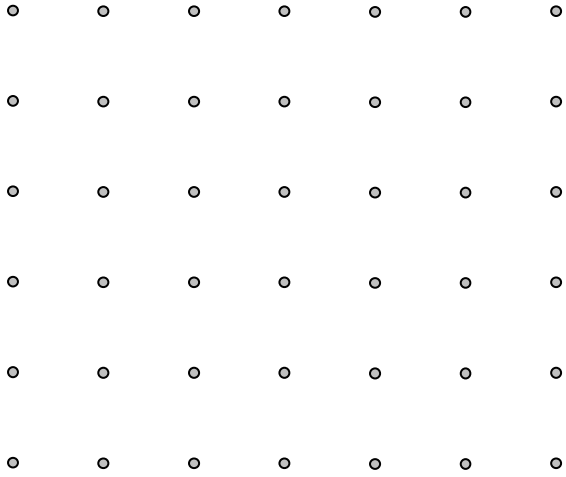
# Refined Unit Cells

We used (620), (260), (444), ( $4\bar{4}\bar{4}$ ), (220) and (440) reflections for refinement



# Summary

- Reciprocal space for epitaxial thin films is very rich.
- Shape and positions of reciprocal lattice points with respect to the substrate reveal information about:
  - Mismatch
  - Strain state
  - Relaxation
  - Mosaicity
  - Composition
  - Thickness ....
- Diffractometer instrumental resolution has to be understood before measurements are performed.



Single crystal

Preferred orientation

Polycrystalline